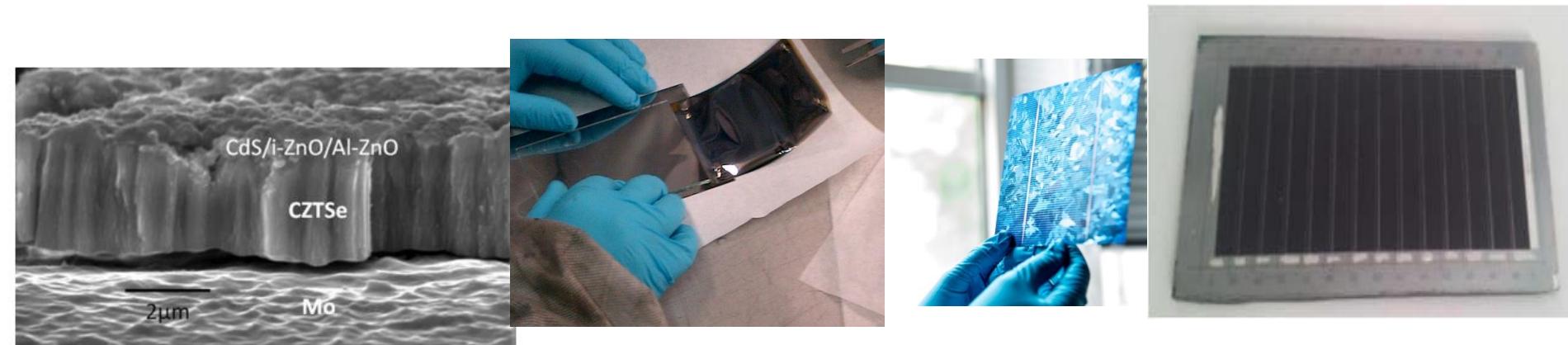


Earth-abundant chalcogenide thin film for PV application: the activity of MIB-SOLAR center at University of Milano-Bicocca

Simona Binetti



The University of Milano-Bicocca

www.unimib.it

S.Binetti, Sydney 27th November 2019



The University of Milano- Bicocca

Founded in
1998
14
Departments
+ 2 Schools
37,197
students

Economics
& Statistics
Law
Education
Sociology
Medicine
Psychology
Sciences

Our Figures & Numbers

981 Faculty Members
721 Employees

28 Specialization Degrees

6 Master Degrees
+12 POSTGRADUATE Programs
(1 year) entirely thought in English

11 double degrees

13 International Summer School Programs

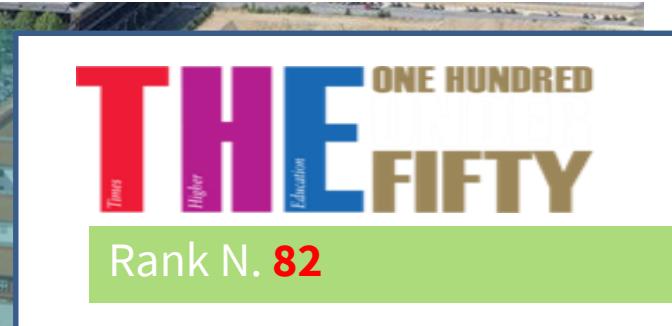
70 total Degree Courses

1 Doctoral School with 19 PhD courses



UNIVERSITÀ DEGLI STUDI
DI MILANO
BICOCCA

2nd in The
National **ANVUR**
ranking on **Big
Universities**



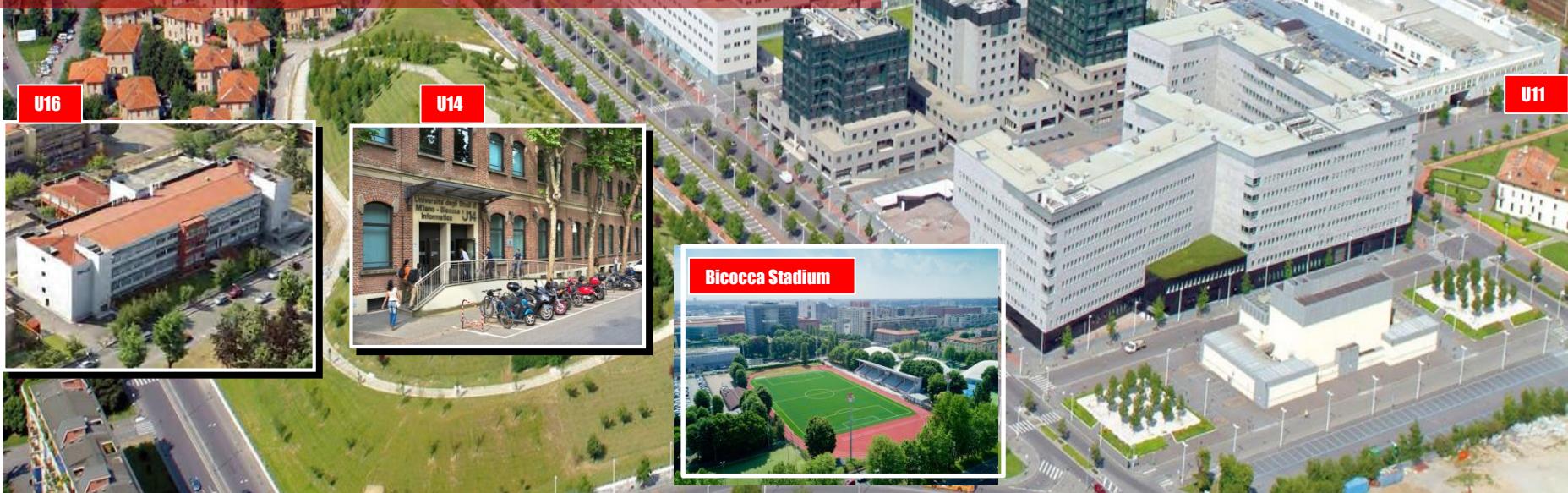
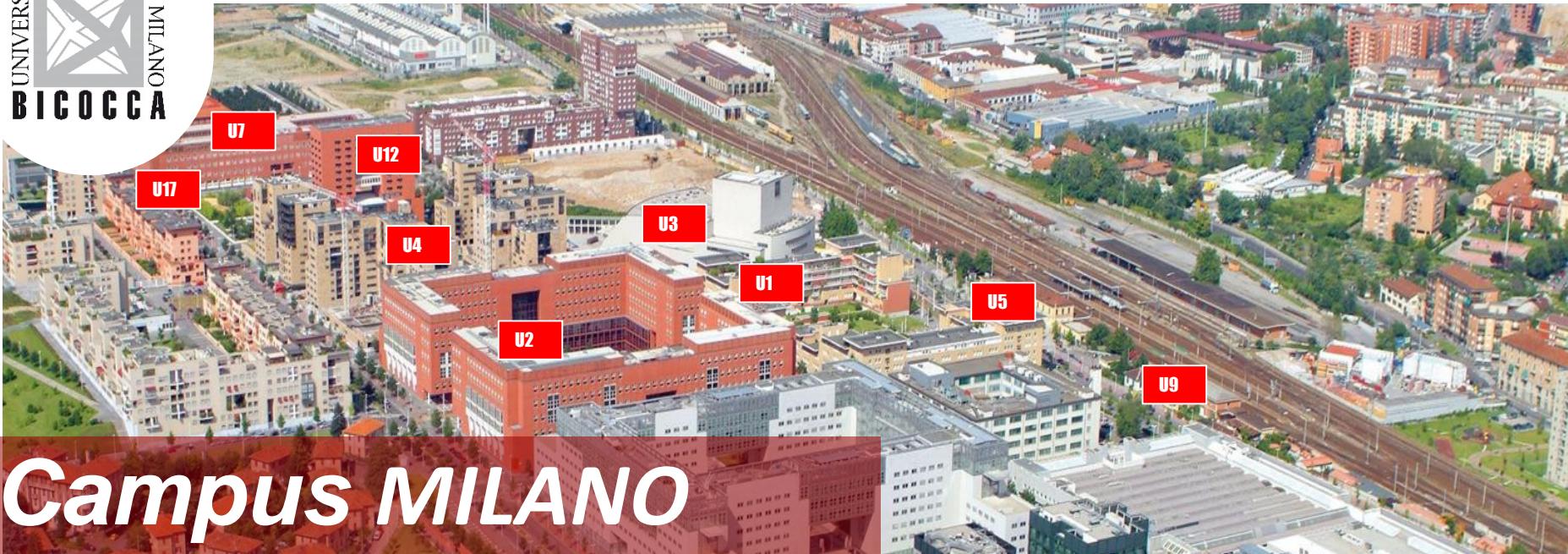
Among universities less
than 50 years' old

What we
achieved in 20
years

8 Dep.ts are
benchmark of
EXCELLENCE

3 in Lombardy for
number of students





Campus Maldives

Marine
Research and
High Education
Center
(#**MaRHE**
Center)

U96



environmental sciences , marine biology, science of tourism , human geography

<http://www.marhe.unimib.it/>



Bachelor and Master

Degrees:

- Materials Science
- Chemistry
- Optical Technologies
- Phd in Materials Science

In a competitive funding from the Italian Government the SdM was granted to the best 180 Italian Departments :
The project “Electrical Power and Energy Vectors from Renewable Sources “ FLEXILAB; research activity on materials for a sustainable energy cycle.

Base
Research

FABRICATION OF CELLS AND MODULES

To support SME
companies:
from Research
to Market

Founded in 2010

fully equipped
synthesis and
characterization
labs

Mission

PV center allowing to pass from
an academic research to the setup of
a prototype devices

100 m²
ISO 7
clean room

FULL MEMBER OF



MIB-SOLAR equipment



Material Characterization

Raman spectroscopy
FT-IR, FT-NIR
HR Scanning Electron Microscopy (EBIC)
X-ray Diffraction
Photoluminescence spectroscopy
UV-Vis/NIR spectroscopy
Hall measurements

Material and cell preparation

CIGS, CZTS and DDSC cells and mini-module
innovative growth process for CIGS
sputtering system
glove box
laser scribing machine
Hotplates, furnaces
screen printers
UV-ozone cleaners

Devices Characterization

solar simulators up to 6 x 6 inches
I/V characterization
Internal and external quantum efficiency
light soaking chamber for cell ageing stability studies
electrochemical impedance spectrometer

MIB-SOLAR RESEARCH TOPICS

Inorganic Photovoltaic devices

- Silicon
- Inorganic thin film (CIGS – CZTS)

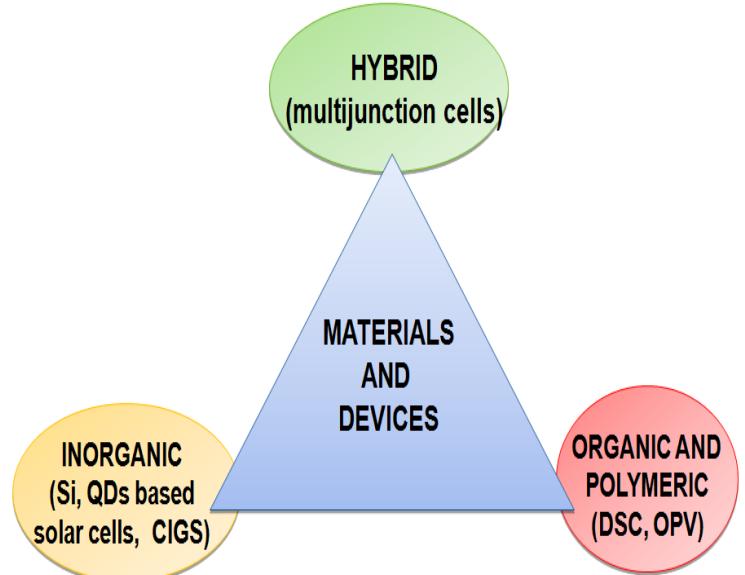
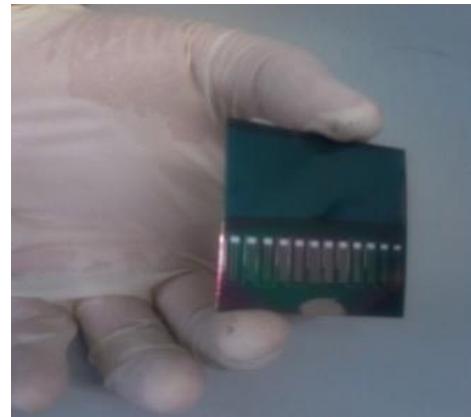
Organic Photovoltaic devices

- Dye sensitized solar cells

Solar fuels (artificial photosynthesis)

- Production of Hydrogen

Purification of Biogas

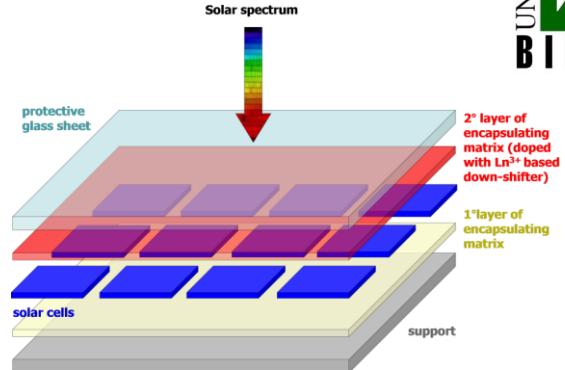


Inorganic Photovoltaic materials and devices

S.Binetti

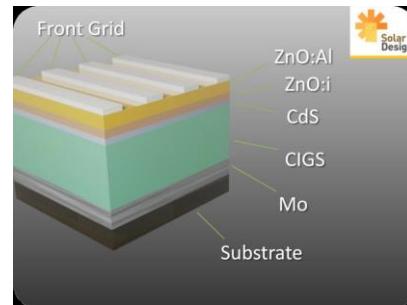
Silicon solar cells

Since 1991 involved in EU project on silicon solar cells
 mc-Si: role of defects (dislocations, grain boundaries)
 Metallurgical silicon : defect and compensation effect
 Light harvesting, (EVA doped with Eu complexes)



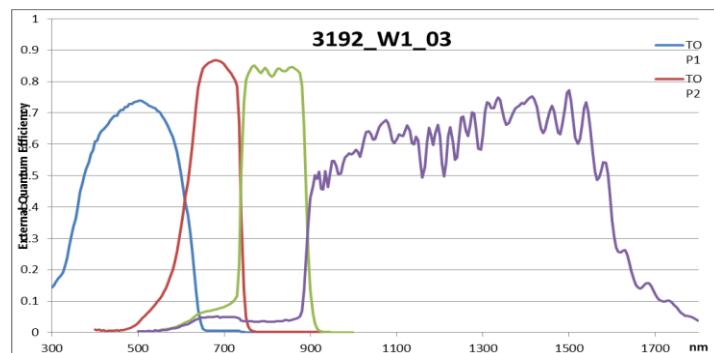
Inorganic thin-film technologies: growth and characterization

- Cu(In,Ga)Se₂
- CZTS
- CMTS



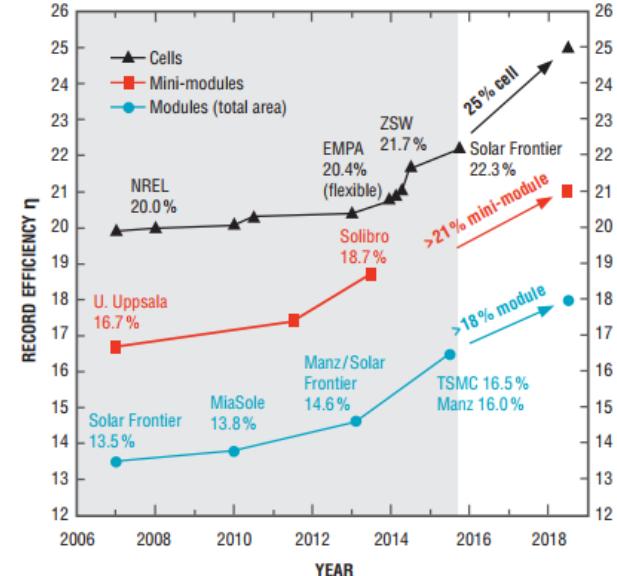
III-V based multijunction solar cells : characterization

- AlInGaP and AlInGaAs for 4 junction devices for space application (CESI Spa) ($h=34\%$)
- Integration of triple junction on silicon



Chalcogenide thin film solar cells

- ✓ **Cu(In,Ga)Se₂ (CIGS)** solar cells are very well positioned in the PV technologies with present record efficiencies 22.9 % for small cells and 16.5 % for production size modules
(Total world-wide CIGS production capacity is ~2 GWp/a)



- ✓ Diversification of production and design of CIGS modules
 - CIGS glass-glass products for BIPV
 - Flexible and light weight CIGS modules for PIPV



<http://cigs-pv.net/white-paper-for-cigs-thin-film-solar-cell-technology/>

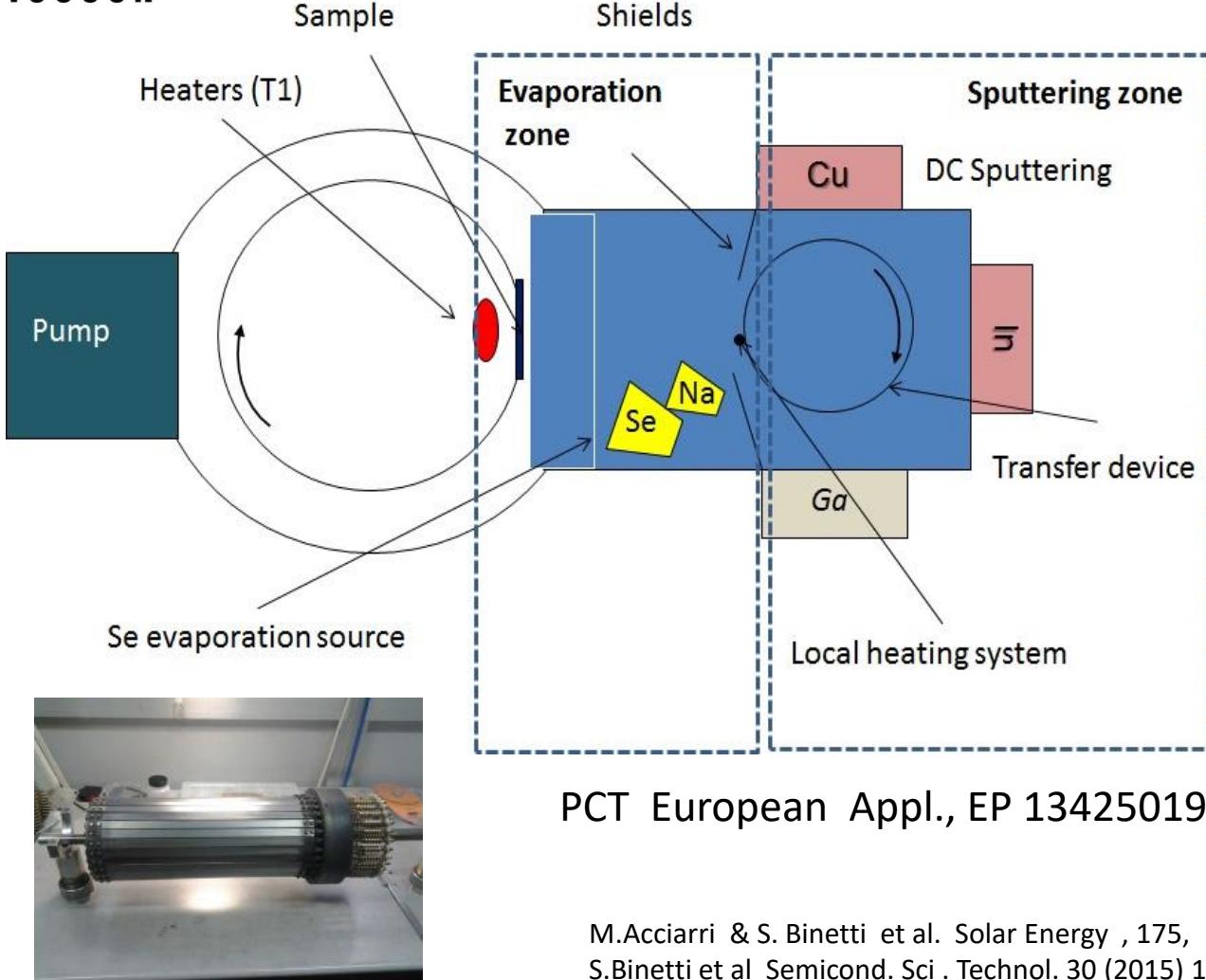
- ✓ CIGS can be used as bottom cell in tandem devices ($\eta > 30\%$)
- ✓ Lower temperature coefficients, higher shading tolerance, a good low light performance are also key CIGS properties, plus a short energy payback time

CIGS technology: open questions

- Complexity of the absorber layer (significant challenges for uniform film properties across a large area)
- the knowledge of the absorber layer is not sufficient
- Current production should increase
- So far being able to produce solar panels at prices that can compete with polycrystalline or cadmium telluride panels has not been possible.
- New deposition system for an easy scale up roll-to-roll configuration (for flexible substrate) is necessary

Most CIGS solar cells are nowadays produced using a co-evaporation technique that involves vacuums and can be costly and time-consuming.

A new hybrid sputtering /evaporation process *



1. The metal precursors are sputtered on rotating transfer devices
2. Then the metals are evaporated on the substrate by local heating elements in a Se atmosphere
3. The sputtering and the evaporation processes continue up to the desired thickness is reached
4. Cooling steps in the presence of Se

PCT European Appl., EP 13425019

M.Aciarri & S. Binetti et al. Solar Energy , 175, 16-24 (2018)

S.Binetti et al Semicond. Sci . Technol. 30 (2015) 105006

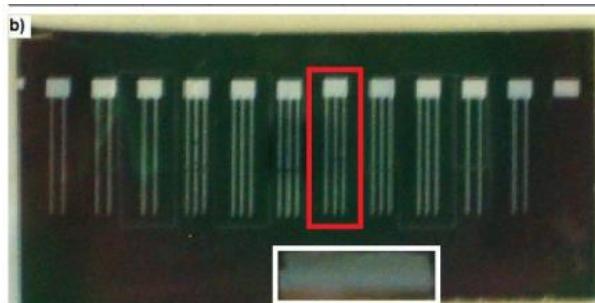
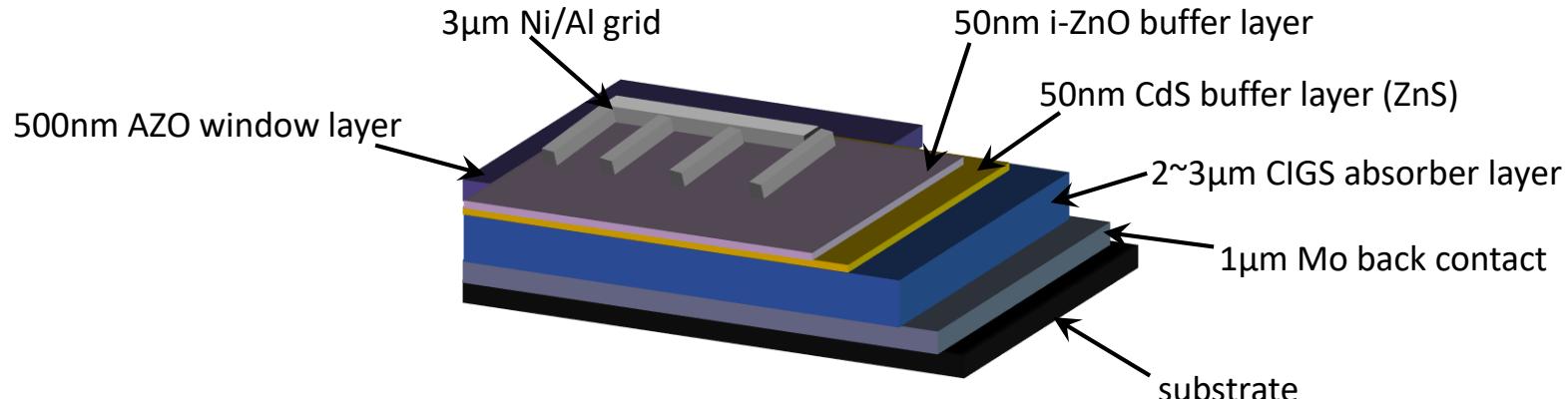
J. Parravicini, et al. Applied Spectroscopy 71(6), 1334-1339 (2017)

J. Parravicini et al. Applied Optics, 57 (8), 1849 (2018)

“ transfer devices made by graphite stripes”

Basic steps of CIGS solar cell process @ MIB-SOLAR

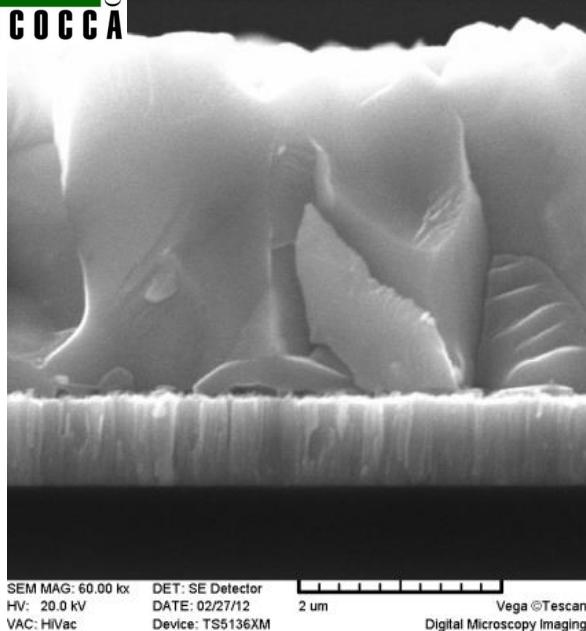
- Deposition time is 30 min (CIGS layer $\approx 2 \mu\text{m}$)
- Deposition Process T = 550 °C (or 450 °C)
- Substrates: maximum size 20 cm x 120 cm in a roll-to-roll configuration
- R&D line that gets the cell and their characterization within less 24 h from deposition



We define the cell size by mechanical scribing cell area equals 0.50 cm^2



CIGS Best results



Glass

T = 550 °C

η [%]:	15 %
Voc [mV]:	581.7
FF [%]	72
Jsc [mA/cm ²]:	34.52
Area [cm ²]:	0.48

Without antireflection coating

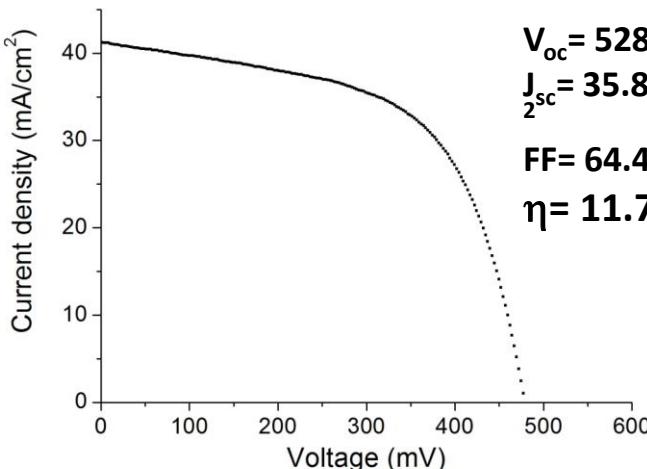
flexible steel foil (120 μm)



η [%]:	13.6
Voc [mV]:	540.6
FF [%]	70.65
Jsc [mA/cm ²]:	31.18
Area [cm ²]:	0.48

Polyimide T = 450°C

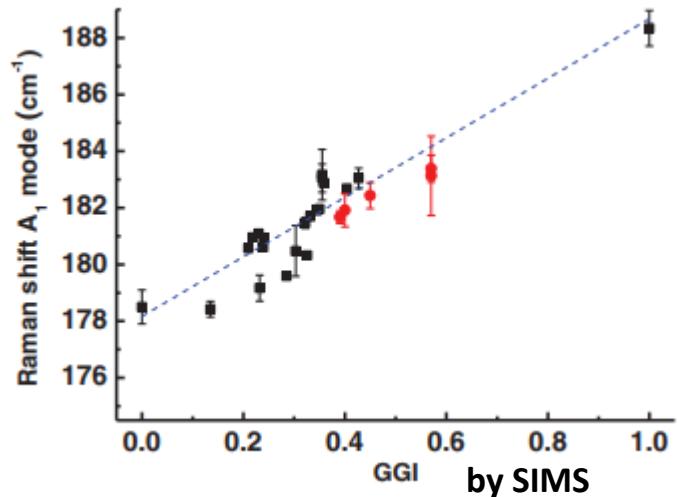
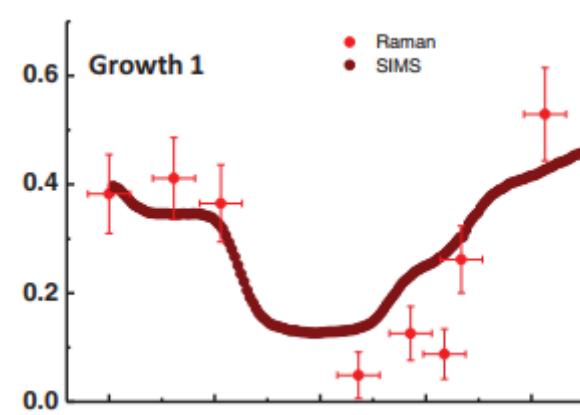
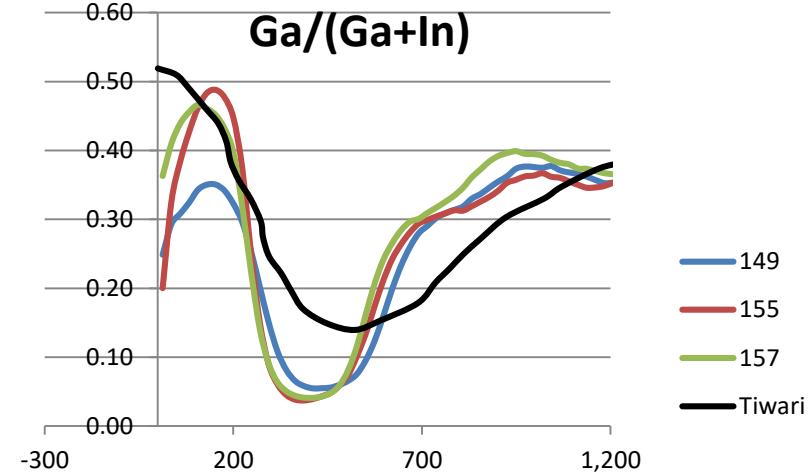
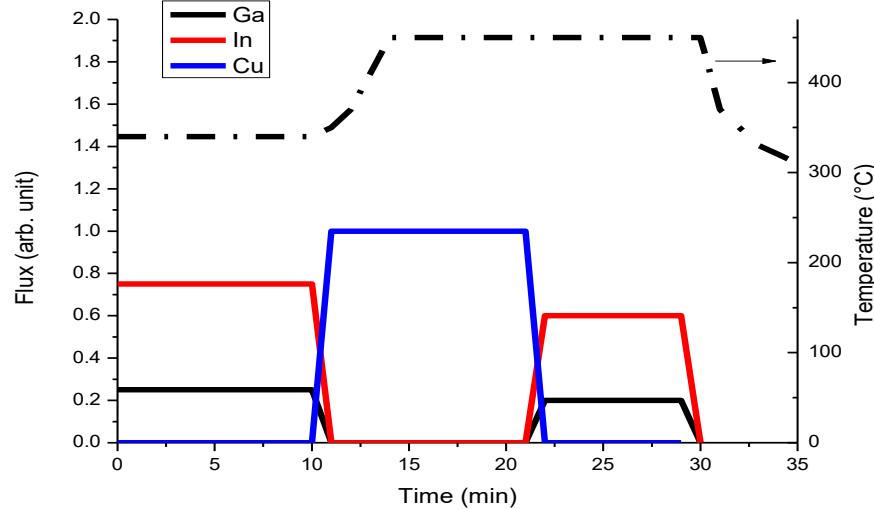
Flexible Thin glass (125 μm)



CuGaS₂
Eg= 2.4 eV
for tandem solar cells



A suitable $[Ga]/([Ga]+[In])$ (GGI) in-depth profile has proved to play a key role in the performance of cells.

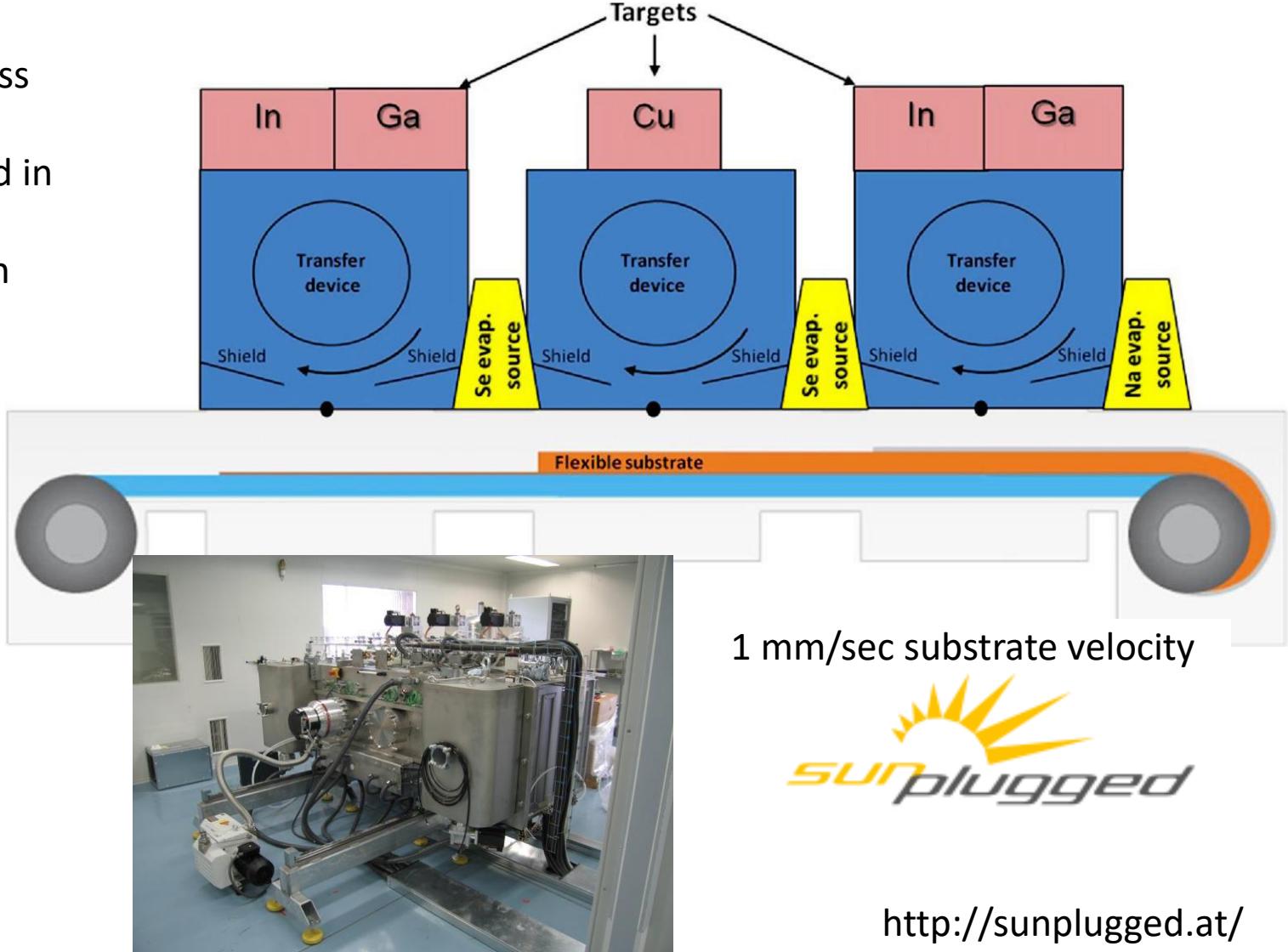


A new method based on repeated bromine etching of CIGS thin film and the measure of the A_1 mode Raman shifts

Applied Spectroscopy 71(6), 1334-1339 (2017)

CIGS deposition system in production line

This process has been transferred in a 1 MW production line



Critical Metals in Inorganic thin film PV Technologies

Abundance in Earth's Crust of the elements

Cu 0.0068 %

Ga = 19 ppm

Zn 0.0078

In= 0.25 ppm

Sn 0.00022 %

Se = 0.05 ppm

Ga= 0.0019%

Se= 5×10^{-6} %

In = 0.00016%

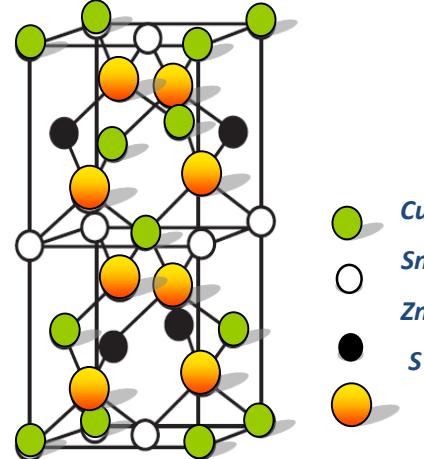
A.Le Donne, V. Trifiletti & S. Binetti "New Earth-Abundant Thin Film Solar Cells Based on Chalcogenides" Frontier in Chemistry 2019
doi: 10.3389/fchem.2019.00297

- ✓ The current indium extraction rate permit to estimate a global CIGS solar module production less than 100 GWp
- ✓ Due to the adverse effects on the environment and human health, the supply and use of cadmium is restricted in Europe under the REACH regulation
- ✓ High price

To raise the competitiveness of thin films based modules, rare and toxic elements should be avoided in all layers of the solar devices.

Strong constraints impose to investigate new materials

Kesterite: Cu₂ZnSnS₄



- Kesterite structure
- (CZTS) Environmentally friendly , low cost
- Intrinsic p-type conductivity (Cu_{Zn} antisite – V_{Cu})
- E_g can be tuned between 1.45 and 1.65 eV (DIRECT) or 0.95-1.05eV
- High absorption coefficient ($> 10^4 \text{ cm}^{-1}$)
- Efficiency record $\eta_{\text{record}} = 11\% * (\text{CZTS}) - 12.6\% (\text{CZTSSe})$

*C.Yan et al. Nature Energy 2019, 3- 764

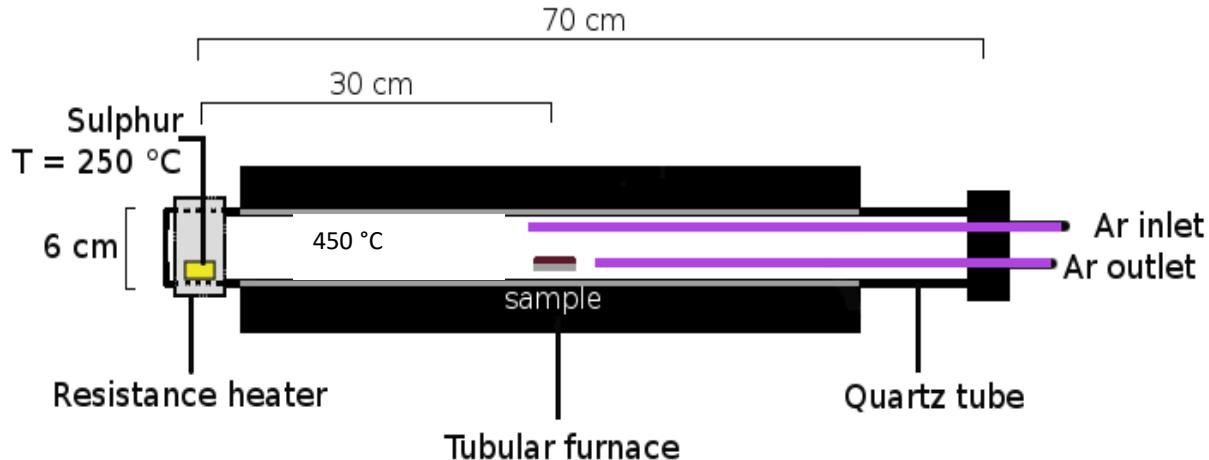
$\text{Cu}_2\text{ZnSnS}_4$ by sputtering

1 . Metal Precursors

-sputtering RF from Cu, Zn, Sn (5N) target on 5x2 cm² Mo coated SLG

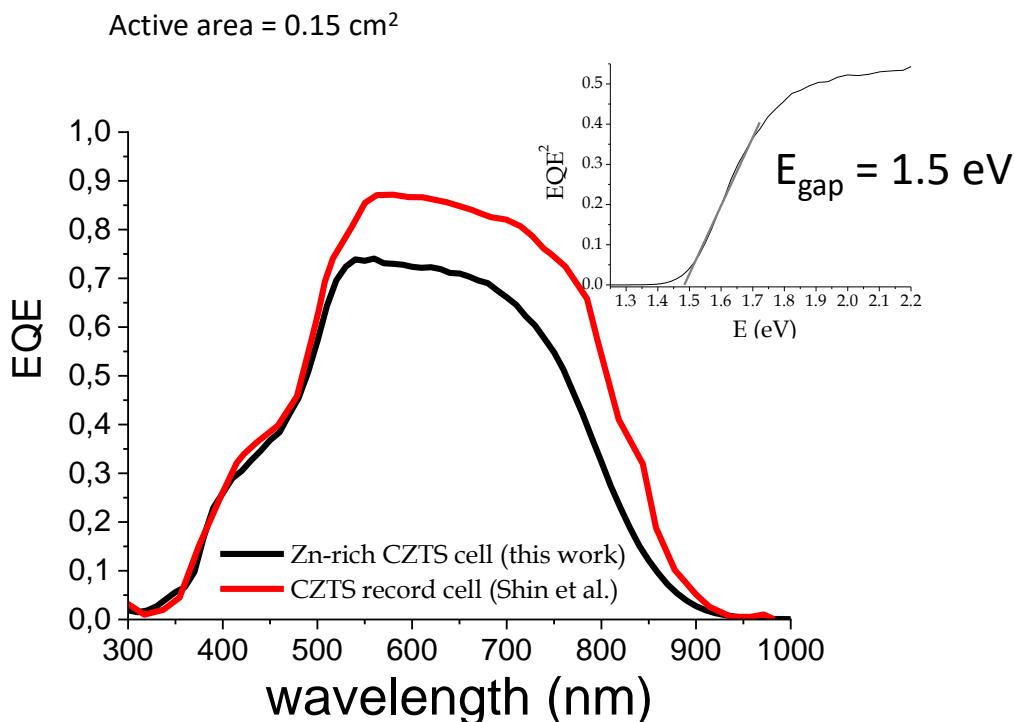
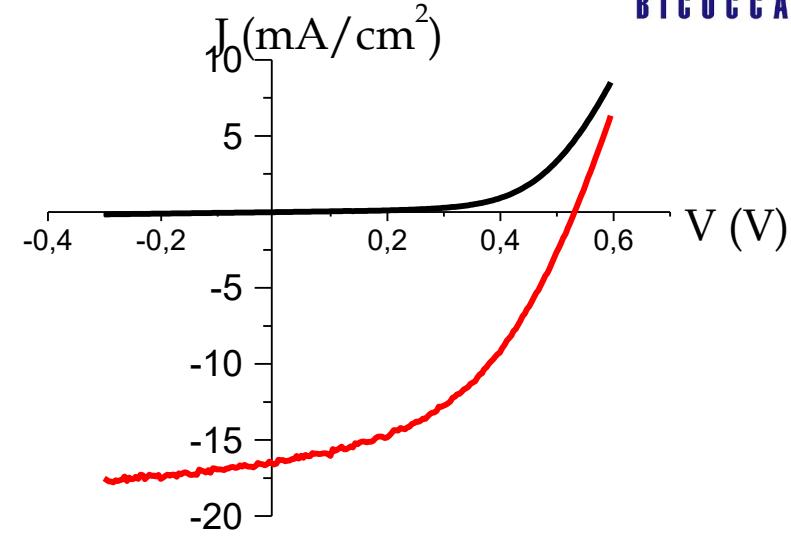
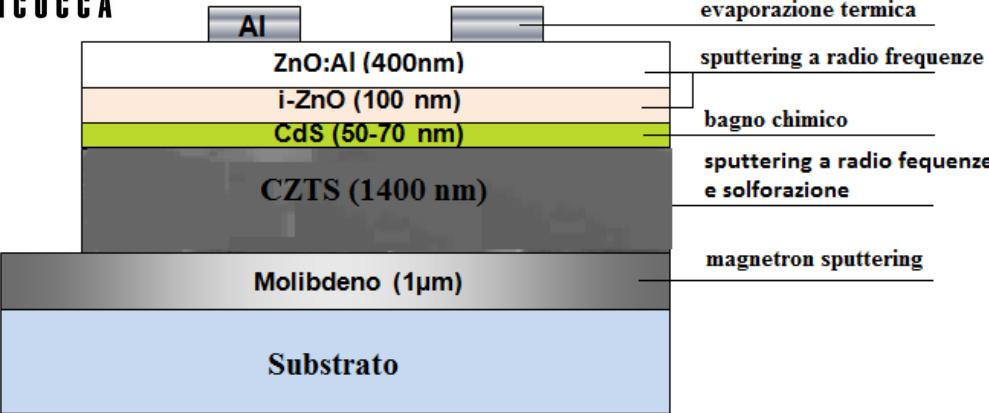
2 . Sulphurization process

0.5 – 0.2 g of S in graphite crucible@250 °C in Ar flow = 30-40 cm³/min - T= 550 °C for 60'



S. Marchionna, P. Garattini, A. Le Donne, M. Acciarri, S. Tombolato & S. Binetti Thin Solid Films **542**, 114 (2013)
A. Le Donne S. Marchionna, P. Garattini, R.A. Mereu, M. Acciarri & S. Binetti International J.of Photonergy (2015)

CZTS by sputtering results

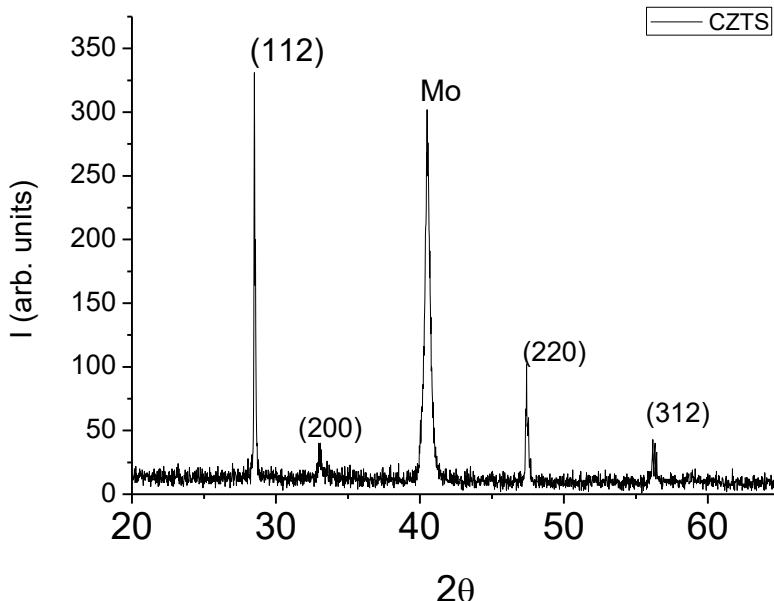


η [%]:	3,95	Voc (mV)	531
FF [%]	44,8	Jsc (mA/cm ²)	16,6

S. Marchionna, et al. & S. Binetti Thin Solid Films **542**, 114 (2013)

CZTS sputtering samples: Material Properties

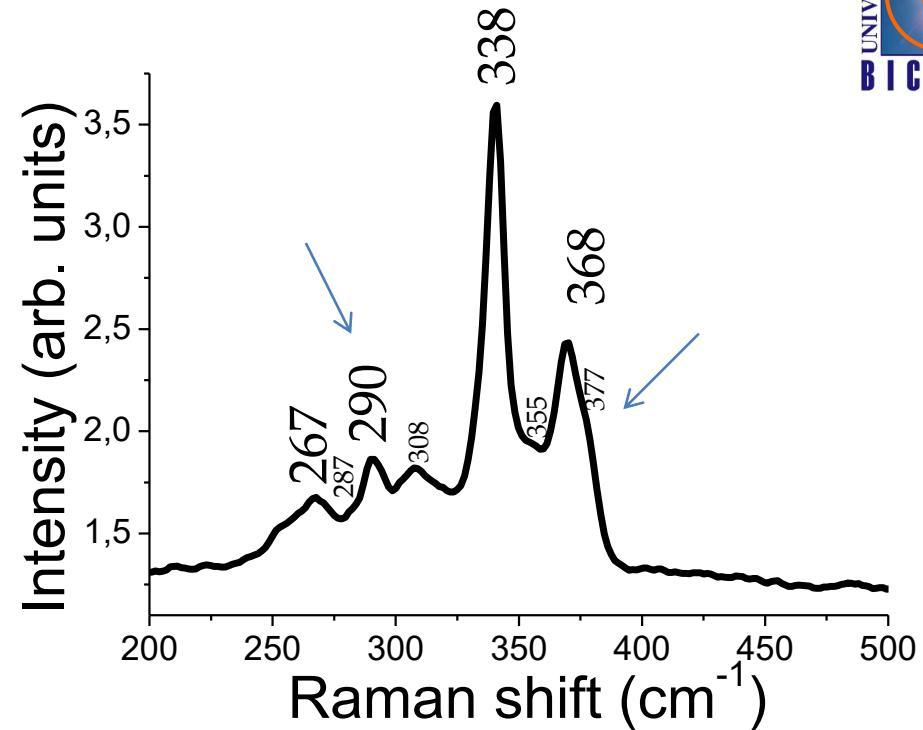
$\text{Cu}/(\text{Zn}+\text{Sn}) = 0.5$
 $\text{Zn}/\text{Sn} = 2.3$



Kesterite Structure

EDX :

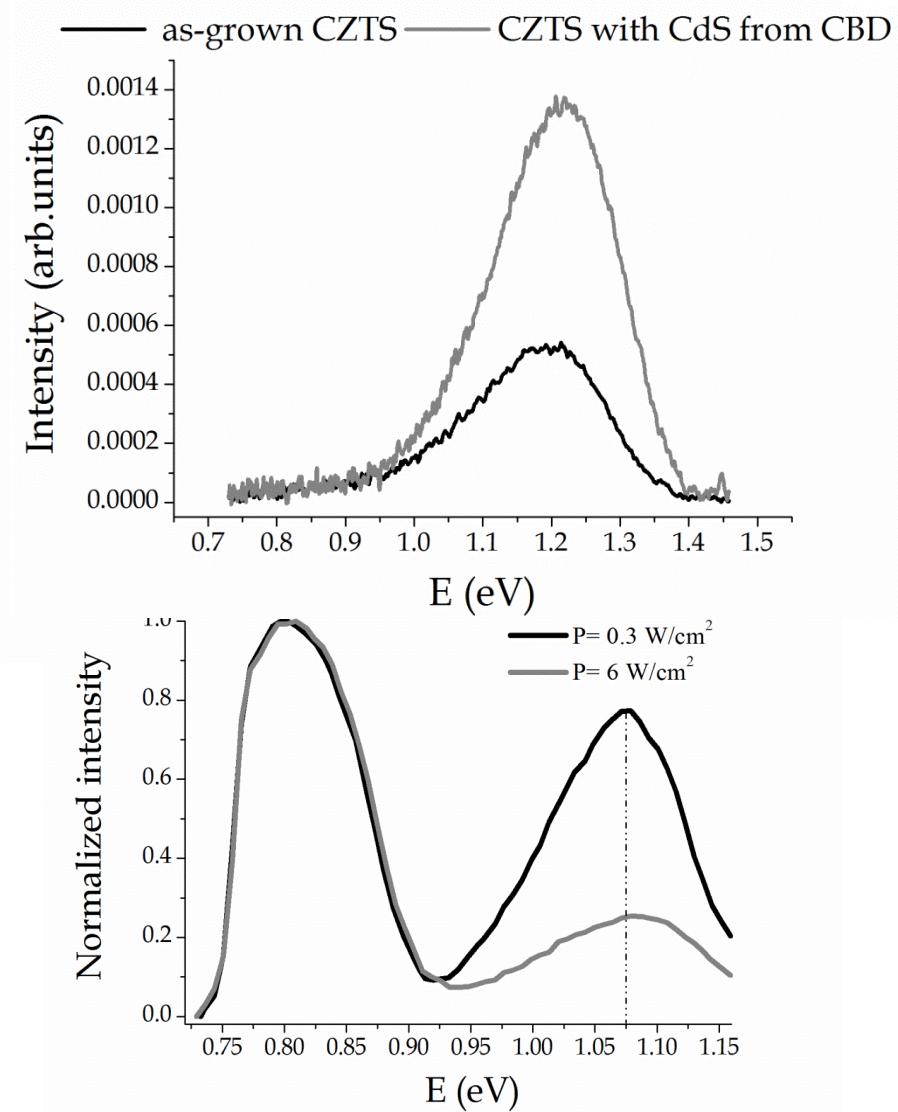
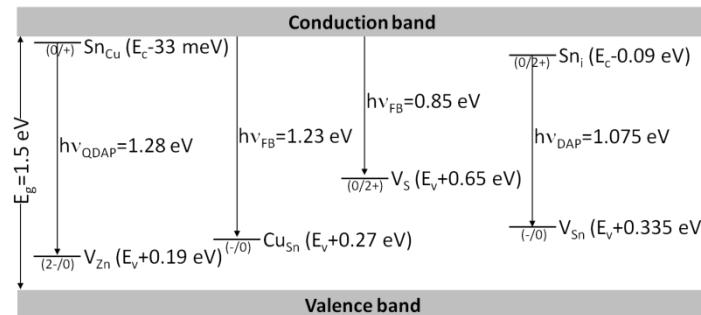
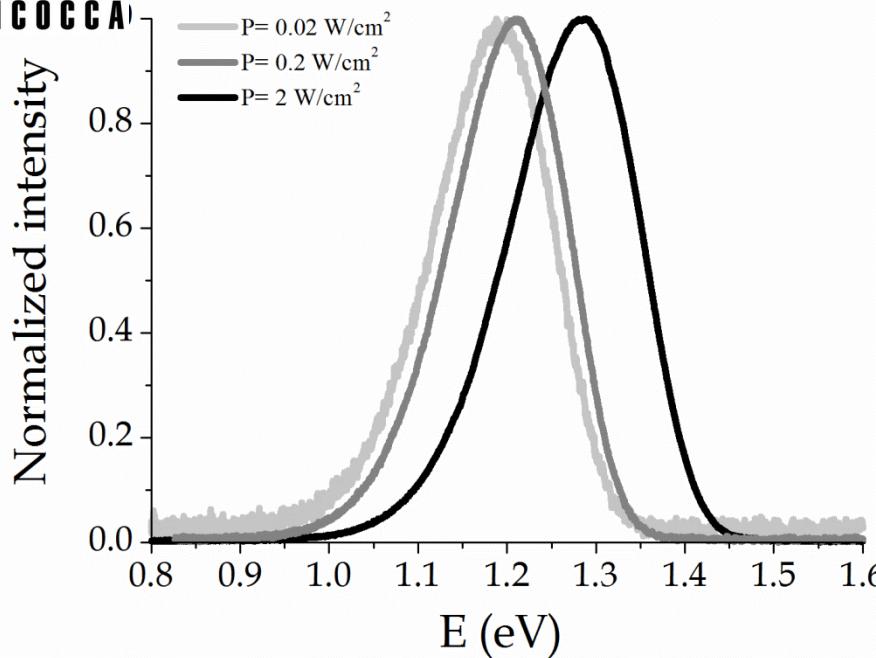
The mean atomic concentration of Cu, Zn, Sn and S resulted 15%, 16%, 10% and 47%,



Along with the CZTS characteristic modes at 267, 287, 338 and 368 cm^{-1} , the Raman spectrum shows additional contributions at 290, 355 and 377 cm^{-1} , typical of cubic CTS

Eventually ZnS spurious phases were removed by etching in HCl (aqueous solution 5% at 75 °C for 300 sec).

CZTS defect identification by PL



A. Le Donne, S. Marchionna, P. Garattini, R.A. Mereu, M. Acciarri and S. Binetti " J. of Photonenergy Volume 2015, Article ID 583058

High quality CZTS thin films by wet process

Aim: Develop a simple , cheap , no toxic process based on the sol-gel technique

Molecular inks :

CZTSsol was prepared by dissolving in DMSO:

- $\text{Cu}(\text{CH}_3\text{COO})_2 \cdot \text{H}_2\text{O}$;
- $\text{SnCl}_2 \cdot 2\text{H}_2\text{O}$;
- $\text{Zn}(\text{CH}_3\text{COO})_2 \cdot 2\text{H}_2\text{O}$.

After complete dissolution, thiourea was added.



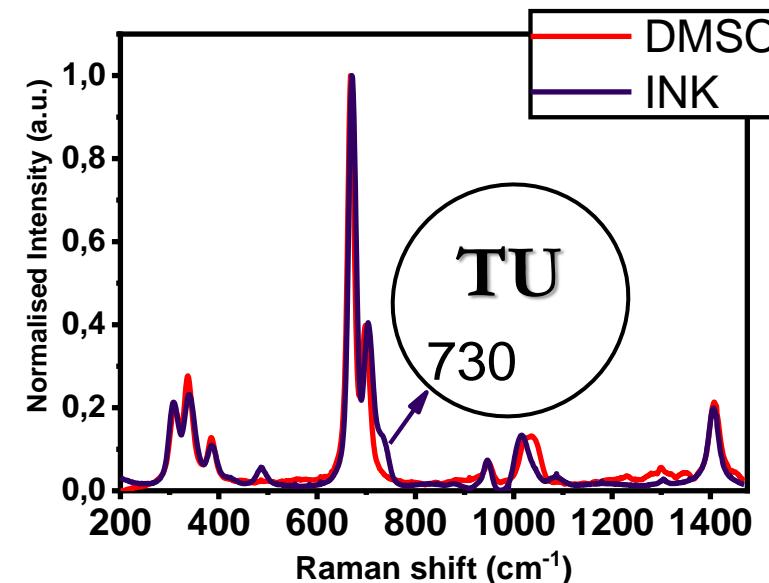
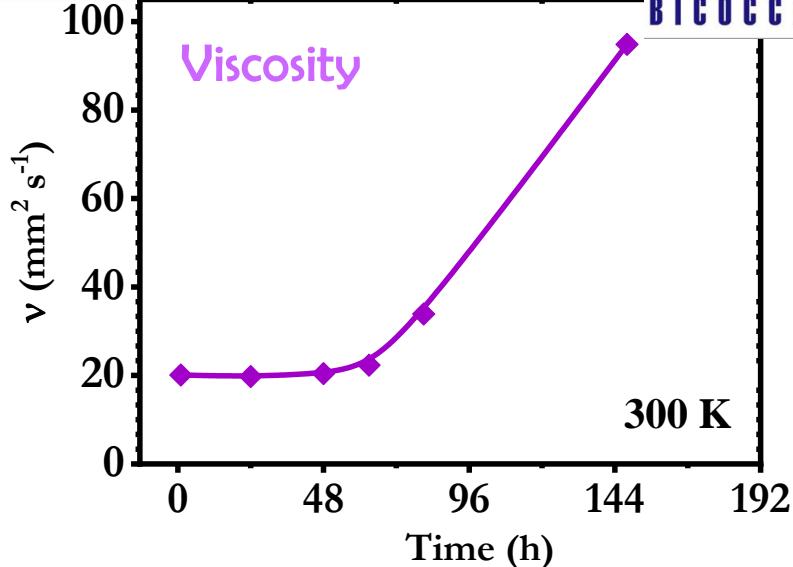
We investigated the composition and stability of the molecular ink

V. Trifiletti et al., Chemistry Select 2019, 4, (17), 4905-4912.

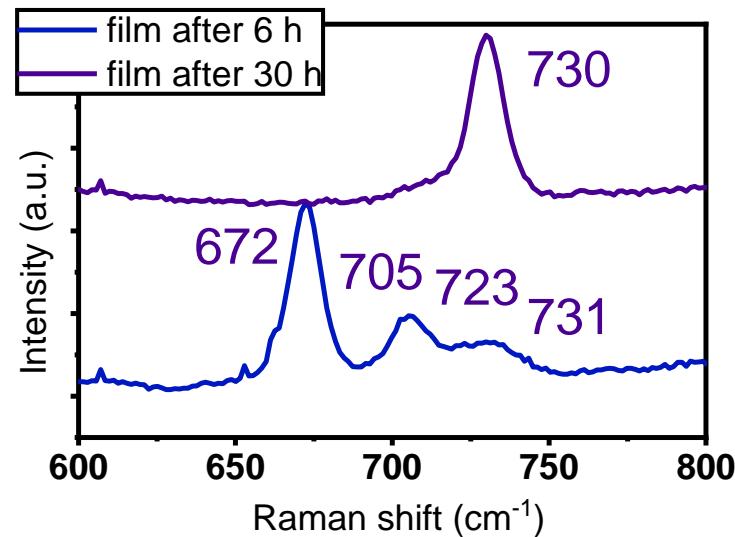
Sol-gel evolution

Ink composition :
 $\text{Cu}(\text{CH}_3\text{COO})\cdot\text{H}_2\text{O}$
 $\text{Zn}(\text{CH}_3\text{COO})\cdot 2\text{H}_2\text{O}$
 $\text{SnCl}_2\cdot 2\text{H}_2\text{O}$
in (DMSO and TU)

V. Trifiletti et al., Chemistry Select 2019, 4, (17), 4905



The Sol-gel viscosity increases due to polycondensation process, until the ink evolves to gel



After drying in air the DMSO signal disappears, and after 30 hours the thiourea signal splits in bands that are assigned to the metals coordinate by TU.



$\text{Cu}(\text{CH}_3\text{COO})_2 \cdot \text{H}_2\text{O}$

0.8 mmol



$\text{Zn}(\text{CH}_3\text{COO})_2 \cdot 2\text{H}_2\text{O}$

0.5 mmol



$\text{SnCl}_2 \cdot 2\text{H}_2\text{O}$

0.5 mmol



Thiourea

9.0 mmol

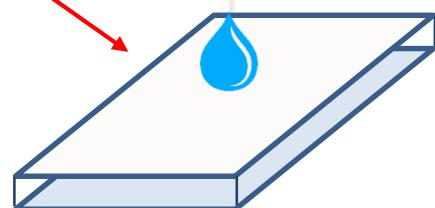
Solvent:
DMSO 3.0 mL



Thin films were fabricated
by
1) a direct drop-casting
of the solution
2) Gelation in 30' at RT
after droplet deposition

loading
1st Layer: 4 $\mu\text{L}/\text{cm}^2$
2nd Layer: 6 $\mu\text{L}/\text{cm}^2$

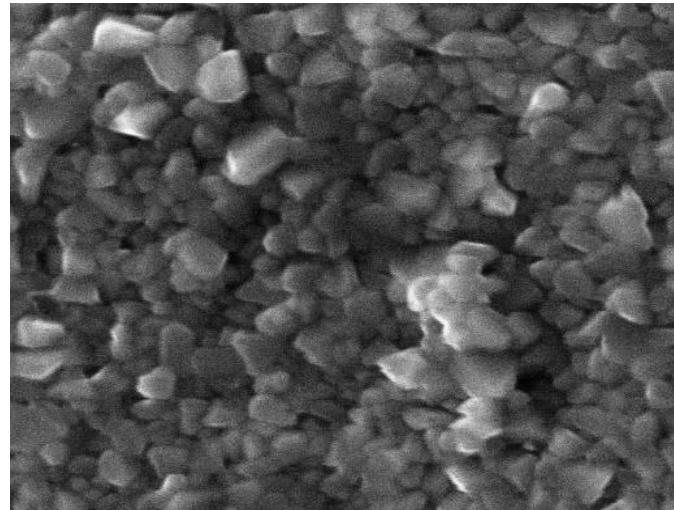
3) 1st and 2nd Layer:
Oven annealing in Argon
@ 550 °C



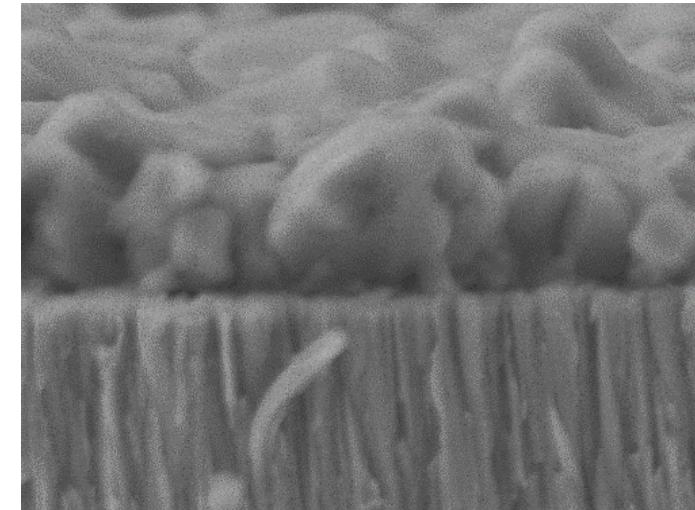
final thickness:
1.2 -1.5 μm

Optimization of the solution/film

Entry#	Cu/(Sn+Zn)	Zn/Sn	Thiourea [conc.]
1	1.00	1	3.7 M
2	0.91	1	3.7 M
3	0.86	1	3.0 M
4	0.83	1	3.0 M
5	0.80	1	3.0 M
6	0.80	1.1	3.0 M
7	0.80	1.2	3.0 M
8	0.80	1.2	2.3 M



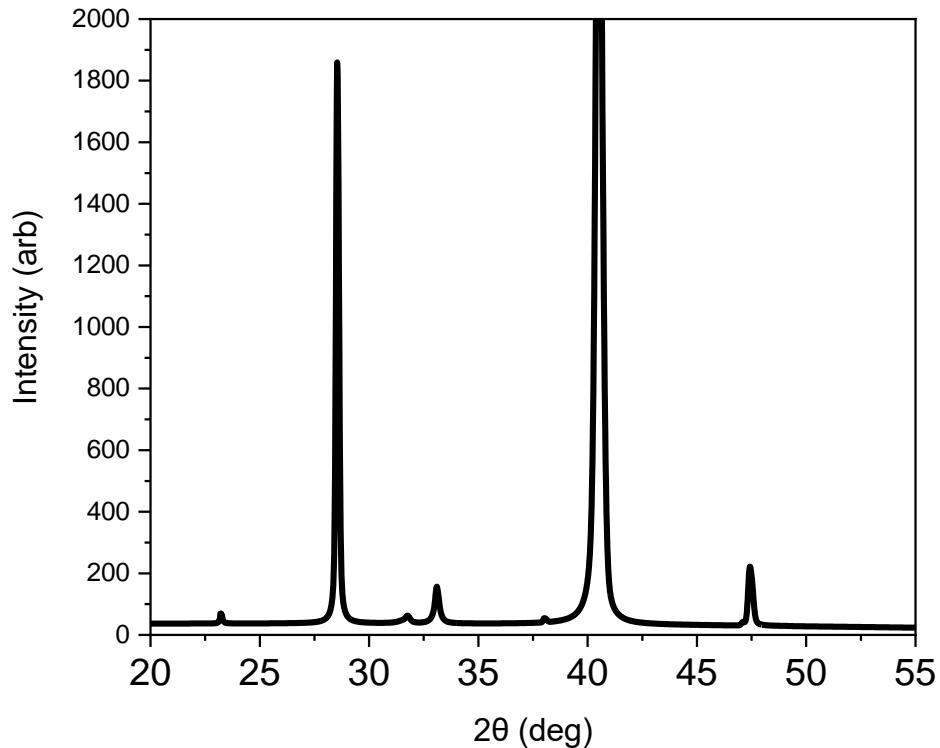
SEM MAG: 30.00 kX DET: SE Detector
HV: 20.0 kV DATE: 09/12/19
VAC: HiVac Device: TS5136XM



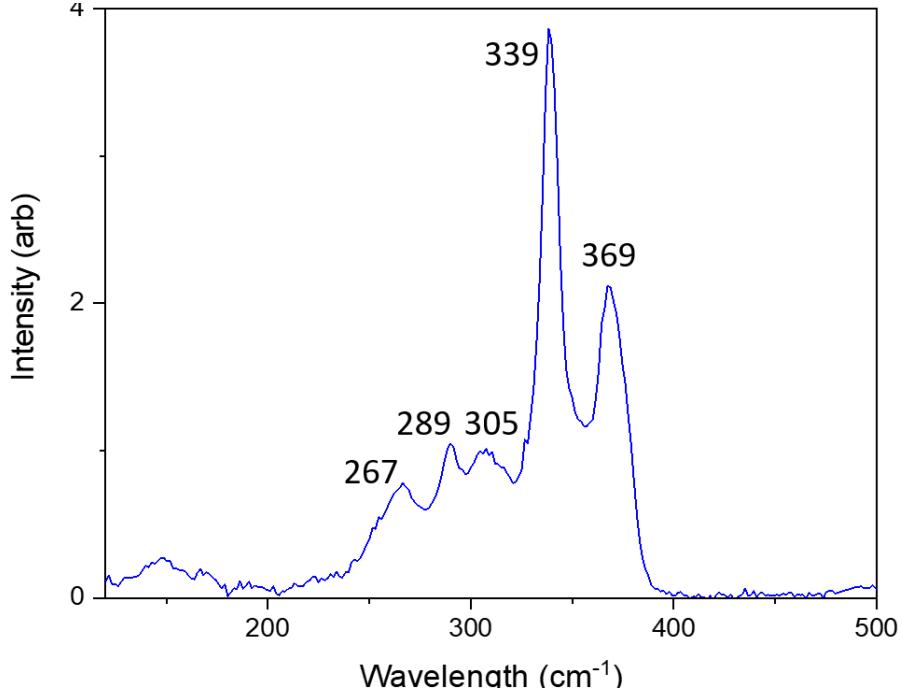
Mag = 100.00 KX 200 nm EHT = 5.00 kV Signal A = SE2
WD = 3.7 mm Date: 12 Sep 2019 ZEISS
Vega ©Tescan
Digital Microscopy Imaging

CZTS drop casting samples: material properties

XRD



Raman Spectra

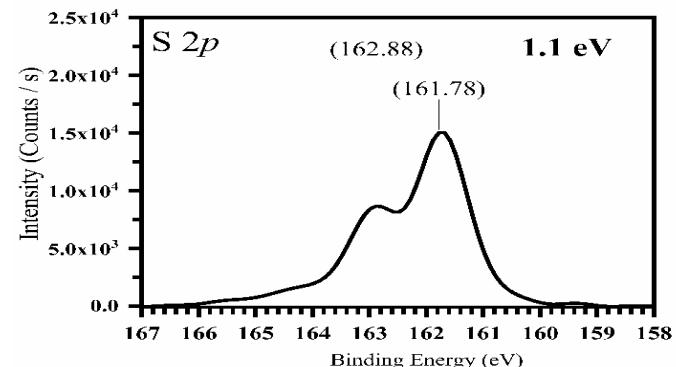
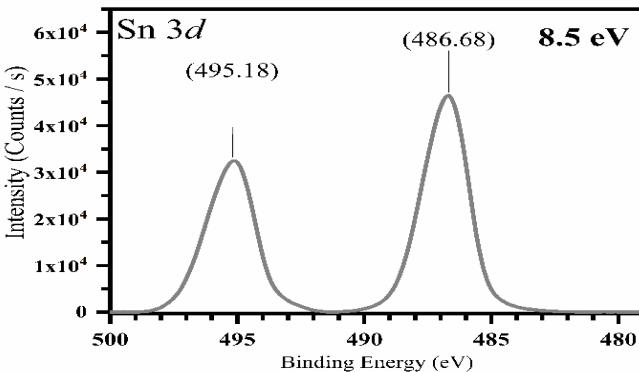
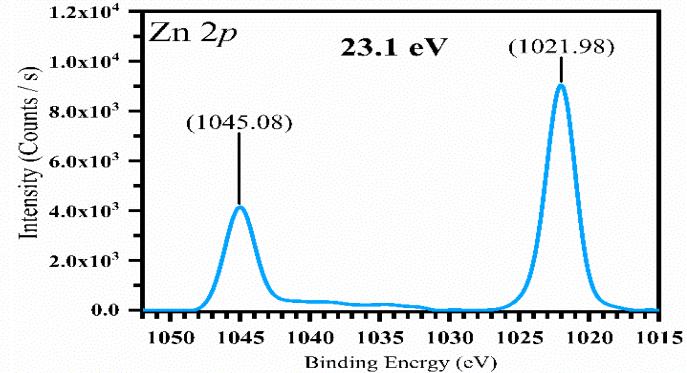
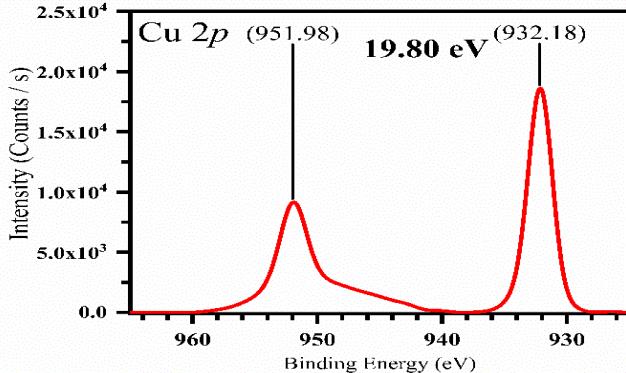


$$2\theta = 16.5, 18.4, 23.3, 28.7, 33.2, 47.5$$

Unpublished results

XPS analysis on CZTS film

From Ms. Sally Luong, Dr Vanira Trifiletti and Dr Oliver Fenwick
School of Engineering and Materials Science ,Queen Mary University of London



Ion Beam Etch:
30 sec x 3 times
(30 nm each time)
Energy 8000 eV
Raster size 1 mm
Cluster size 1000

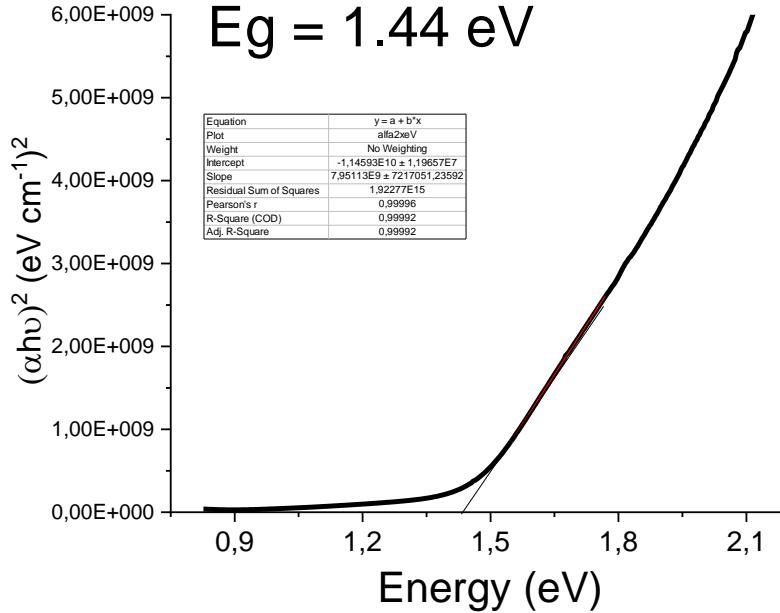
Cu, Zn, Sn, and S oxidation states: Cu (I), Zn (II), Sn (IV) and S (II)

Sample #2

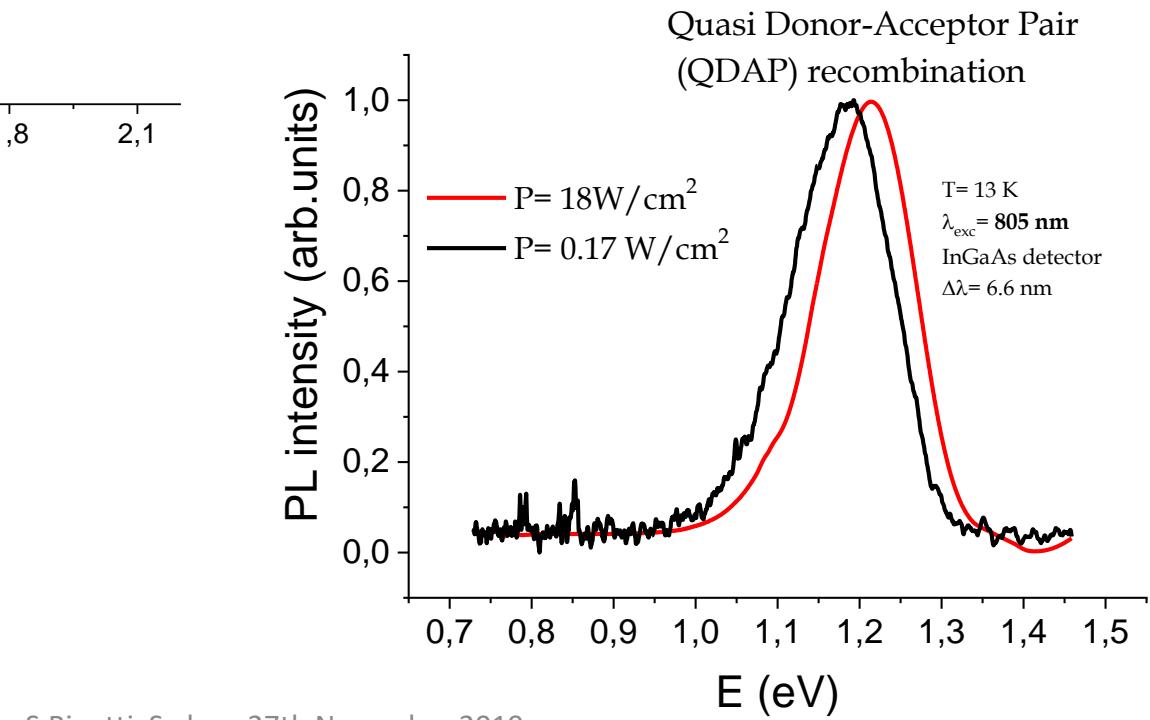
Unpublished results

S.Binetti, Sydney 27th November 2019

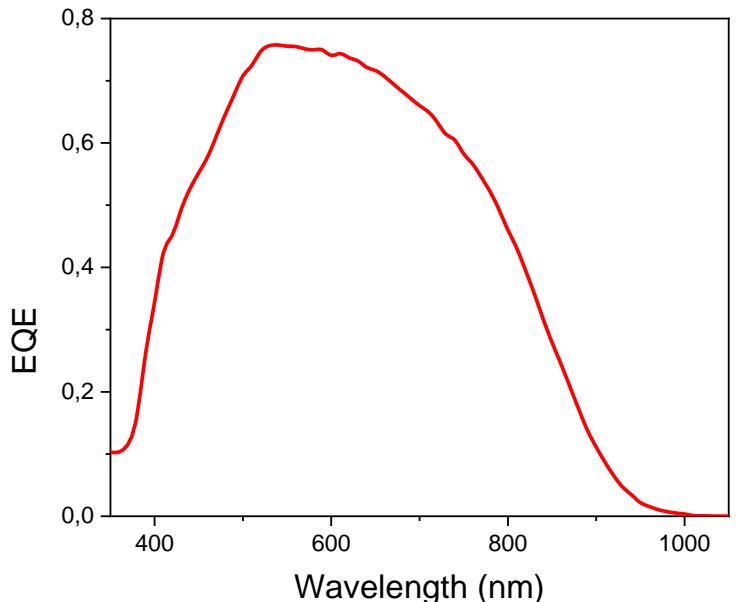
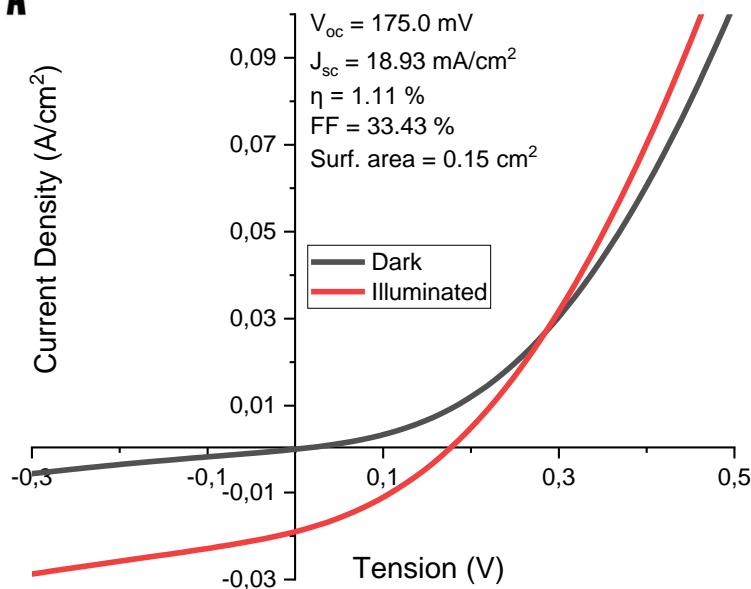
Band gap and PL



Unpublished results



Device Performance



$V_{\text{oc}} = 175.0 \text{ mV}$



Modest V_{oc}

$J_{\text{sc}} = 18.93 \text{ mA}/\text{cm}^2$



Respectable J_{sc}

$\eta = 1.11 \%$

$\text{FF} = 33.43 \%$

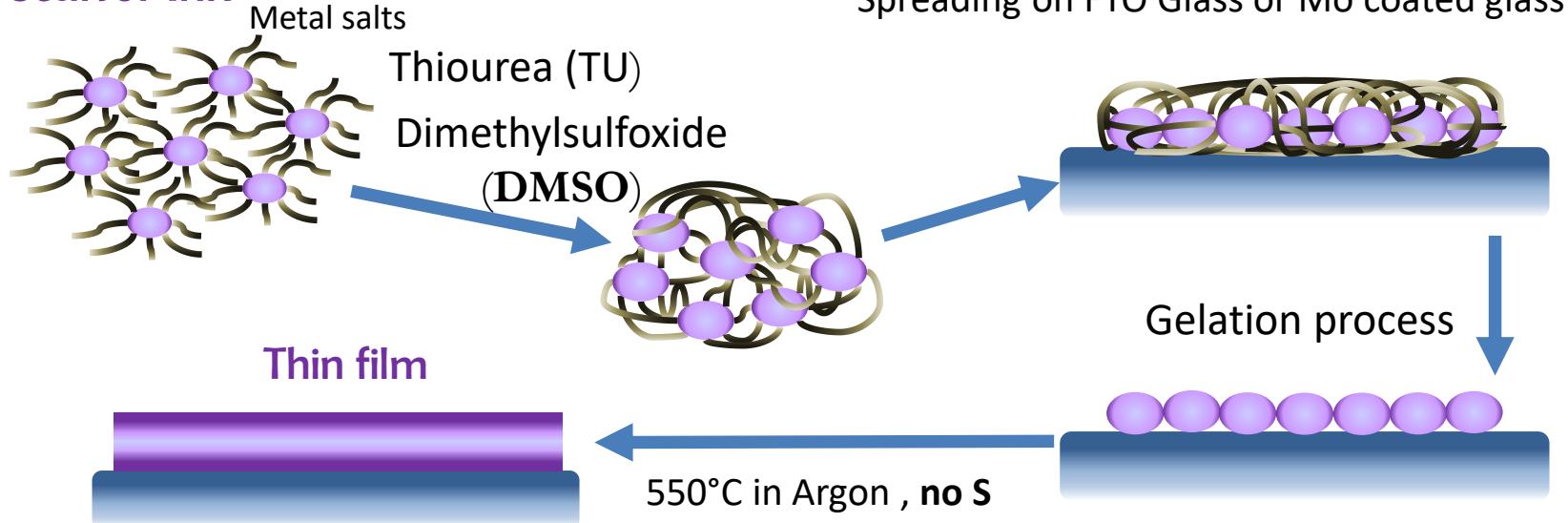
Unpublished results

SCAPS software simulation indicates problems at the interface with the back contact

Kesterite thin films by non toxic solution process

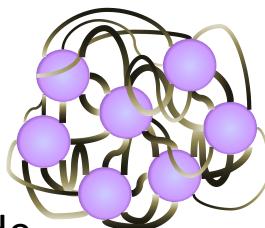
Several deposition methods

1 Precursor Ink

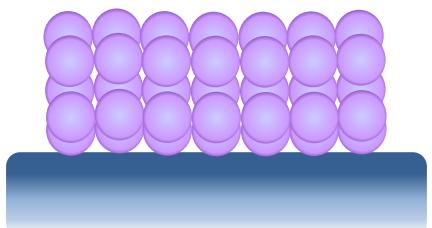


2 Precursor-ink

Metal salts
Thiourea
Dimethylsulfoxide



Spin Coating on substrates
on **FTO or Mo Glass**)



Thermal treatment
without
sulfurization



3 Precursor-ink + a ink jet printer (in progress)

V. Trifiletti et al., Chemistry Select, 2019

Tuning the gap



Molecular inks :

CZFTS sol was prepared by dissolving in DMSO:

- $\text{Cu}(\text{CH}_3\text{COO})_2 \cdot \text{H}_2\text{O}$;
- $\text{SnCl}_2 \cdot 2\text{H}_2\text{O}$;
- $\text{Zn}(\text{CH}_3\text{COO})_2 \cdot 2\text{H}_2\text{O}$.
- $\text{Fe}(\text{CH}_3\text{COO})_2$

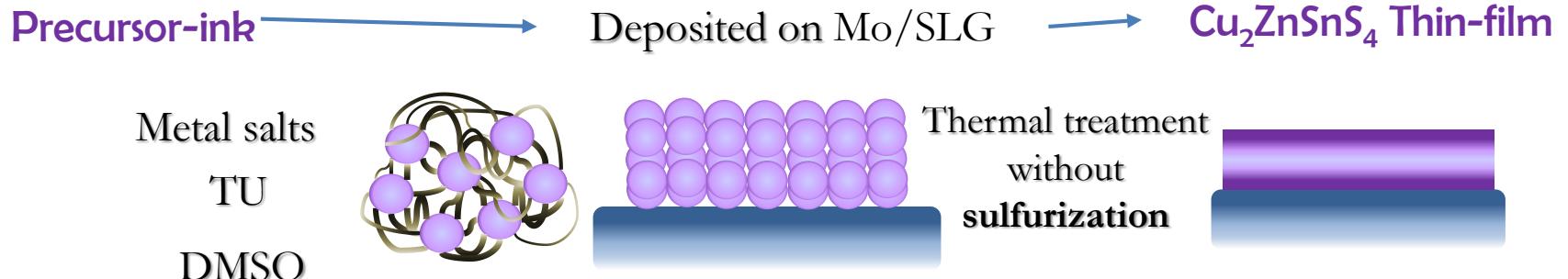
After complete dissolution, thiourea was added.



Ink is spread on Mo SLG and HT @550 °C in S
Preliminary thin film results very promising



CZTS by wet process : summary



In the precursor-ink:

- ✓ TU acts as a monodentate ligand for the metal ions and DMSO solvates them;
- ✓ DMSO supports TU coordination.

Kesterite phase formation in the final film is supported by:

- the acetate groups, which bridge the different metal ions, creating a network, and favour the sol-gel formation;
- the homogeneous distribution of the components.
- Promising material quality and I_{sc}

Open questions

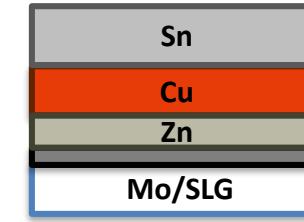
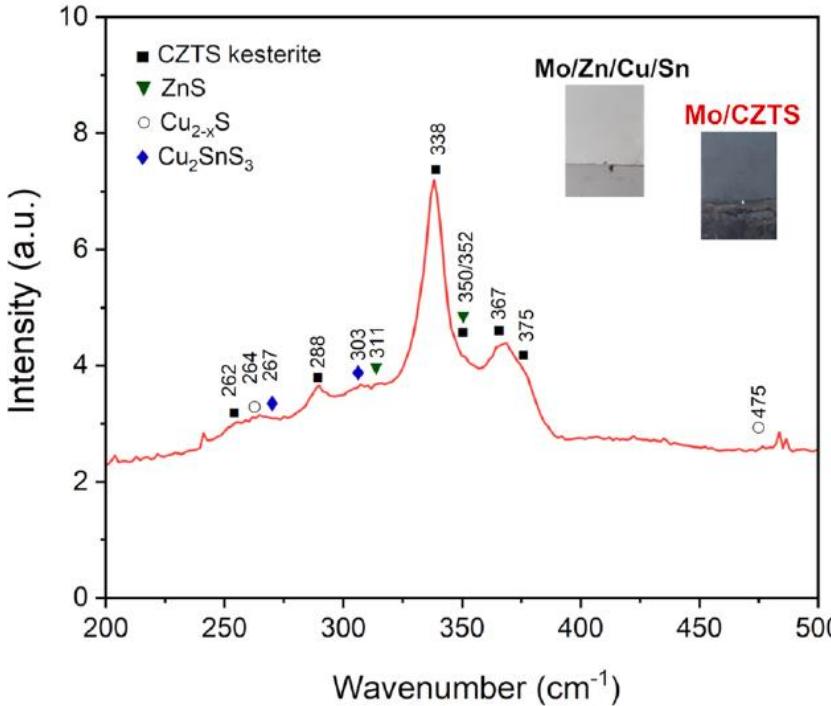
Low efficiency (problem with back contact (MoS_2 ?), no etching with KCN, intermediate passivation layer , alternative buffer layers...)

CZTS by electrodeposition

is one of the most attractive fabrication routes:
large area and low-cost process and easily scalable



stack elemental layers approach of CZT precursors + sulfurization



Advantages :

- Fast steps
- Control on stoichiometry (layer thickness)
- Scalable to large surfaces
- Constrained stacking order

- Lower quality than vacuum processes

non-aqueous plating solution:

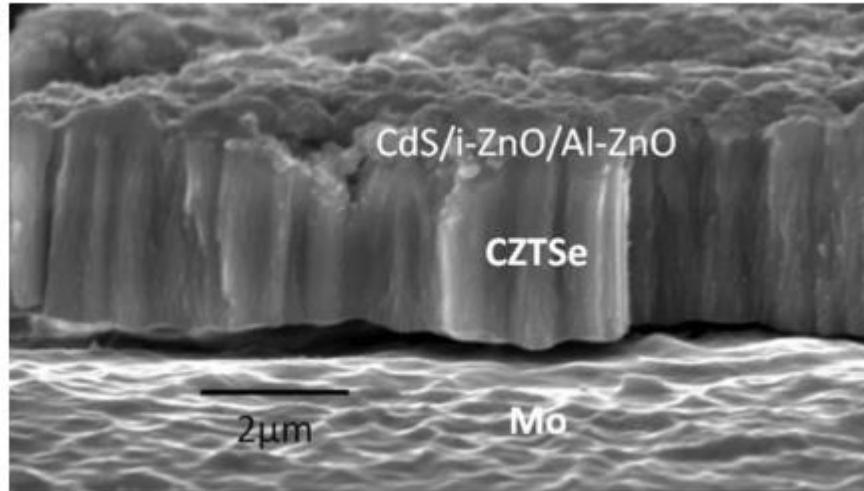
Anhydrous ethylene glycol ; copper acetate 0.05 M ; sodium acetate 1 M ;
diethanolamine 0.8 M; dimethylamine borane complex 0.1 g/L

$\text{Zn/Sn} = 1.1$ $\text{Cu/Zn+Sn} = 0.85$

CZTSe by electrodeposition



Co-electrodeposited Cu-Zn-Sn precursor + sulfurization or selenization on Mo flexible foil which acts both as a substrate for the electrodeposition process and as a back contact



0.1% with $J_{sc} = 3.9 \text{ mA/cm}^2$,
 $V_{oc} = 119 \text{ mV}$ in our first attempt.

M.I.Khalil, et al .“Co-electrodeposition of metallic precursors for the fabrication of CZTSe thin films solar cells on flexible Mo foil” Journal of The Electrochemical Society, 164 (6) D302-D306 (2017)

ALTERNATIVES TO CZTS: $\text{Cu}_2\text{M}^{(\text{II})}\text{M}^{(\text{IV})}\text{S}_4$

with M(II) = Zn, Mn, Fe, Ni, and M(IV)= Si, Ge, Sn



- A new alternative:
 - p-type semiconductor fully based on Earth-abundant and low-cost elements :
 - the abundance in the Earth's crust of Mn is two order of magnitude higher than that of Zn (1100 ppm vs 79 ppm)*,
 - is definitely cheaper (the amount of Zn produced in 2015 was 4'600'000 tons lower than that of Mn (13'400'000 Zn tons vs 18'000'000 Mn tons).
 - Lower Wp cost
 - Up to last year studied as Dilute Magnetic Semiconductor

*A. Le Donne, V. Trifiletti , & S.Binetti * Frontier in Chemistry 2019

Vacuum approach: Metal precursors evaporation followed by annealing in elemental sulfur vapors

Metal Precursors:
4-sources EB
evaporation system
+ **Sulfurization**

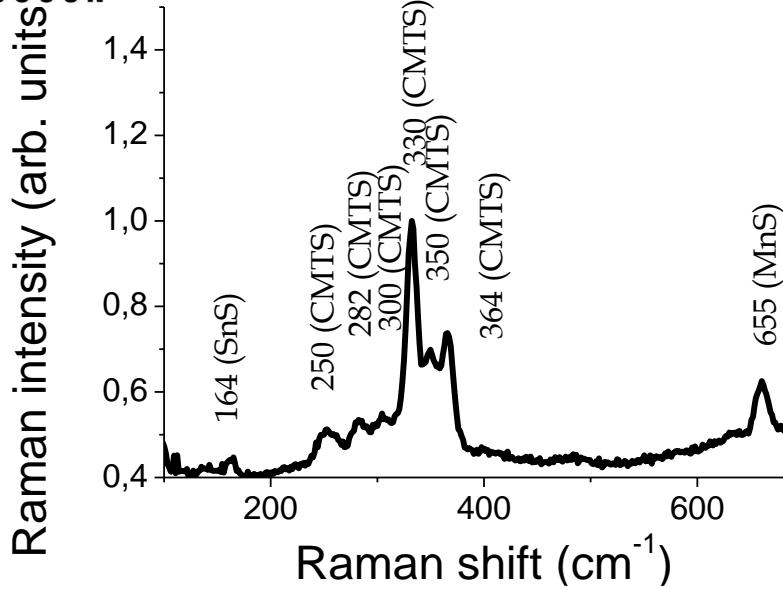
Standard stack structure	Double stack structure	Sandwich stack structure
 Mn Cu Sn	 Mn Cu Sn Mn Cu Sn	 Mn Cu Sn Mn

- Testing :
- the thickness and order of the metal precursors in the evaporated stack
- Annealing temperature : 500°C, 525°C, 555°C, 585°C, for 1 h (ramping rate:15°C/min)
- Pre-annealing to enhance metal intermixing(115°C for 1 h).

Substrates: soda lime glasses coated with sputtered Mo 1 μm

Deposition rate: [Sn] = 0.25 nm/s, [Cu] = 0.12 nm/s, [Mn] = 0.3 nm/s

$\text{Cu}_2\text{MnSnS}_4$ thin film: main properties

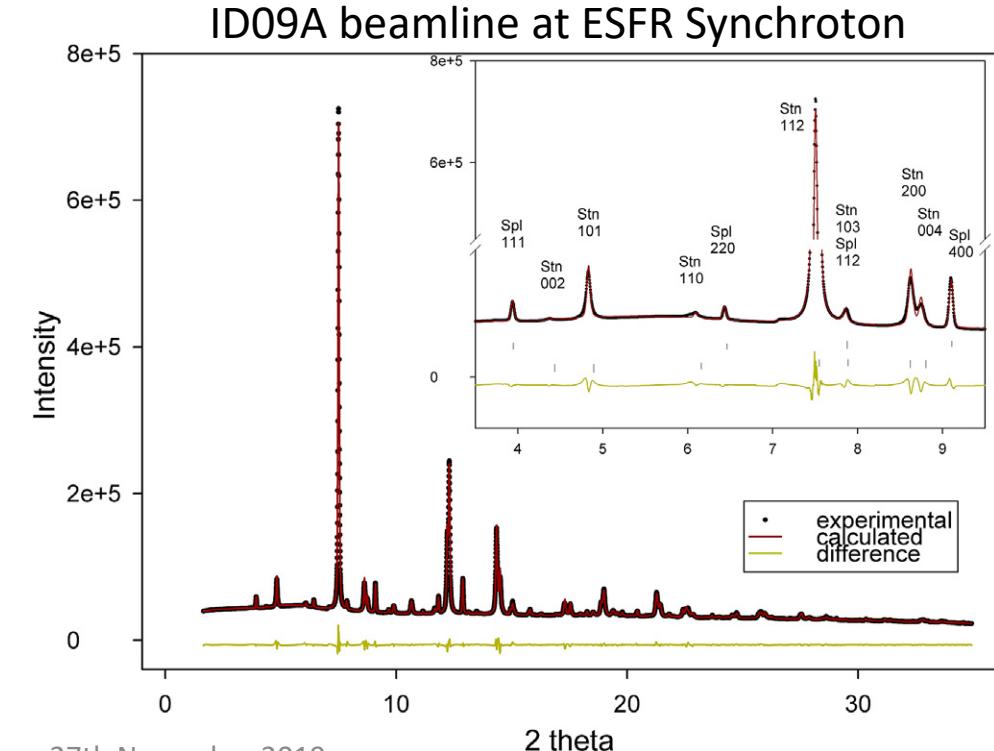


- ✓ Stannite Structure
(+ sulfide compound with spinel structure)

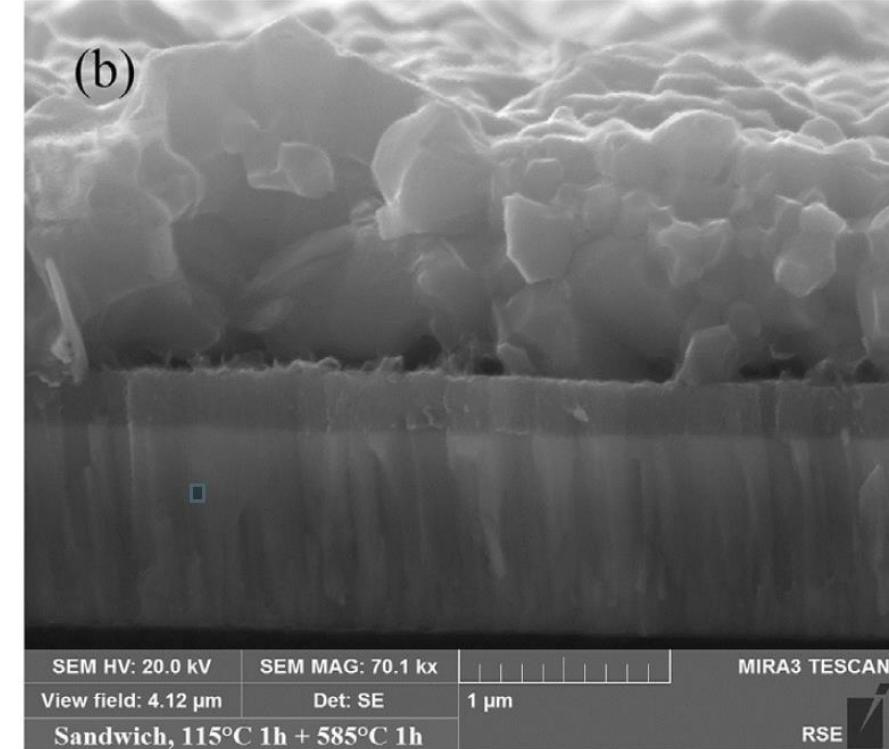
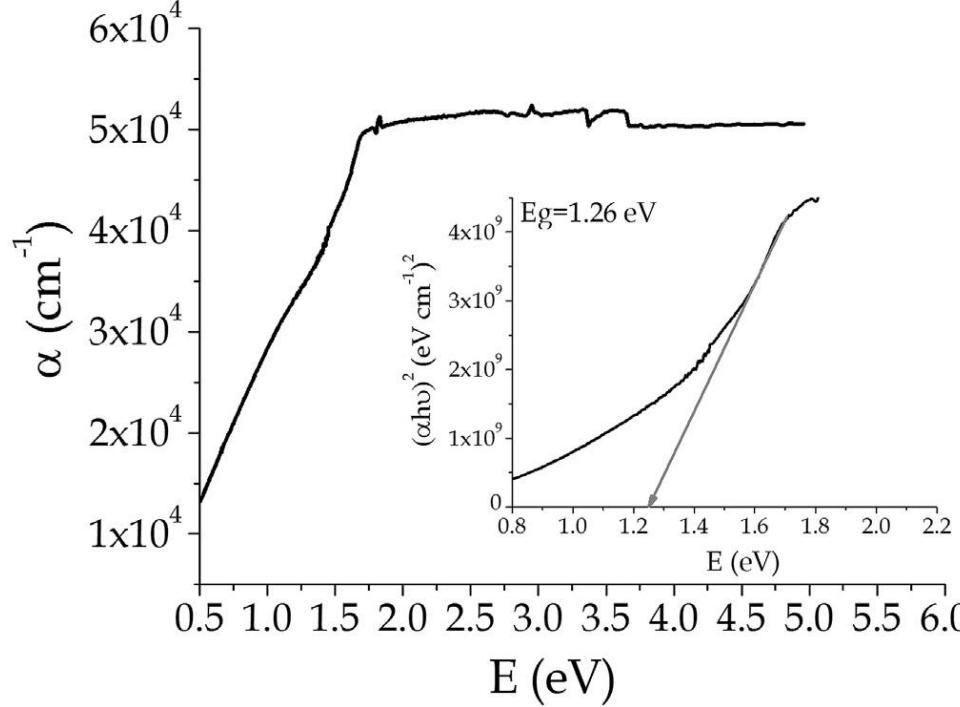
S. Marchionna et al . J.of Alloys and Compounds 693
(2017) 95

- ✓ EDX and Raman analyses confirm that the **CMTS phase** was obtained

- ✓ Lower content of insulating MnS secondary in the case of lower Mn layer thickness (i.e. 135 nm)

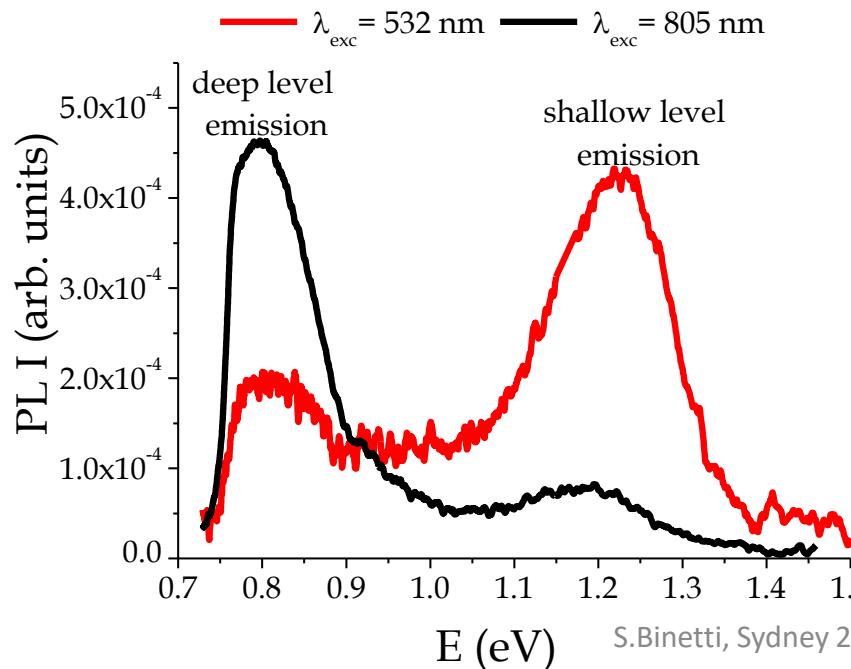
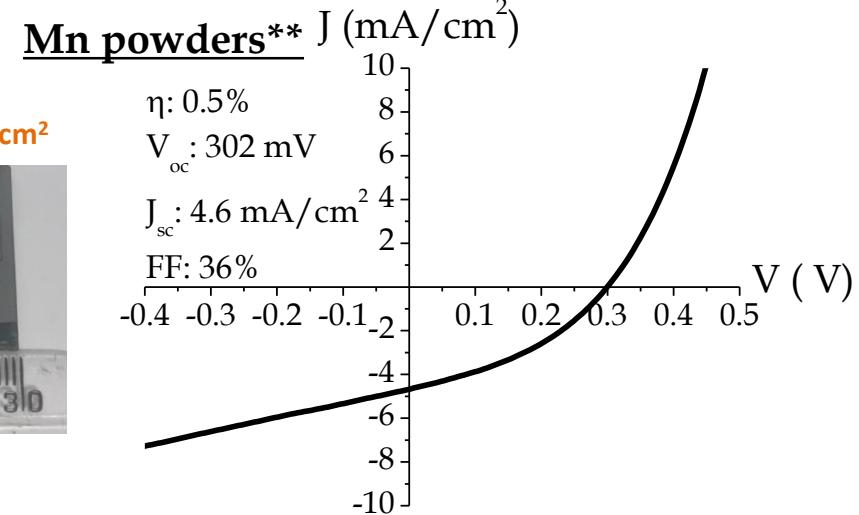
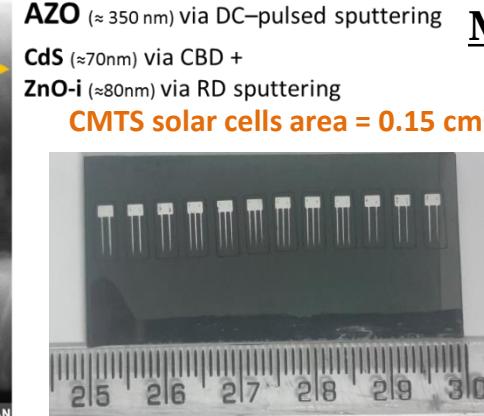
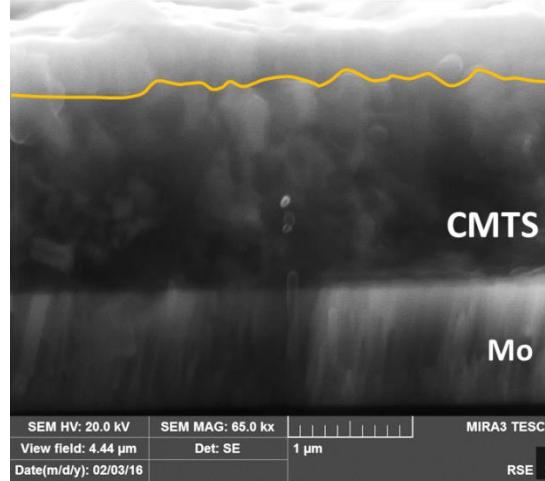


$\text{Cu}_2\text{MnSnS}_4$



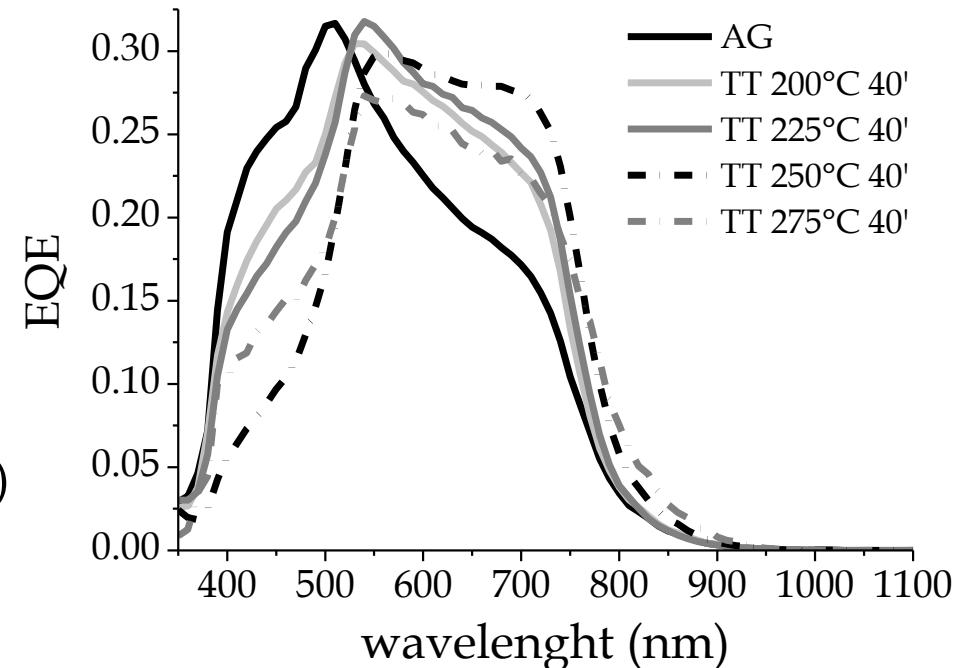
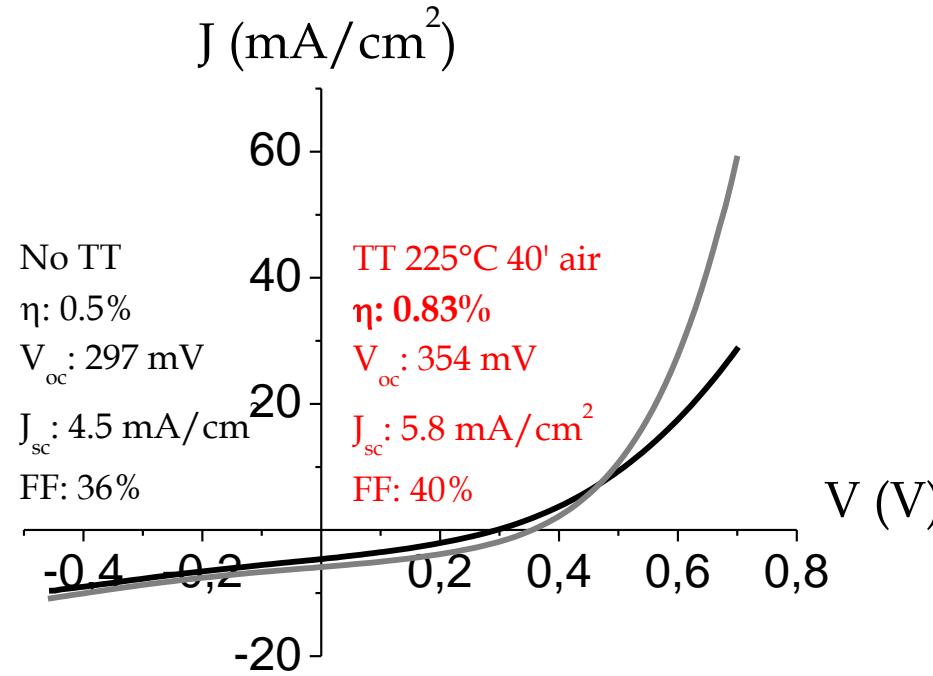
- ✓ high absorption coefficient (**$5 \times 10^4 \text{ cm}^{-1}$**) and direct band (**1.26 eV**) suitable for PV applications have been obtained

Cu₂MnSnS₄: solar cell prototypes



- ✓ Deep recombination center responsible for the emission at **0.8 eV** is associated to a **bulk defect**.
- ✓ Secondary phases (sulfurization ramping rates to be reduced)

Effect of low temperature annealing on PV performance

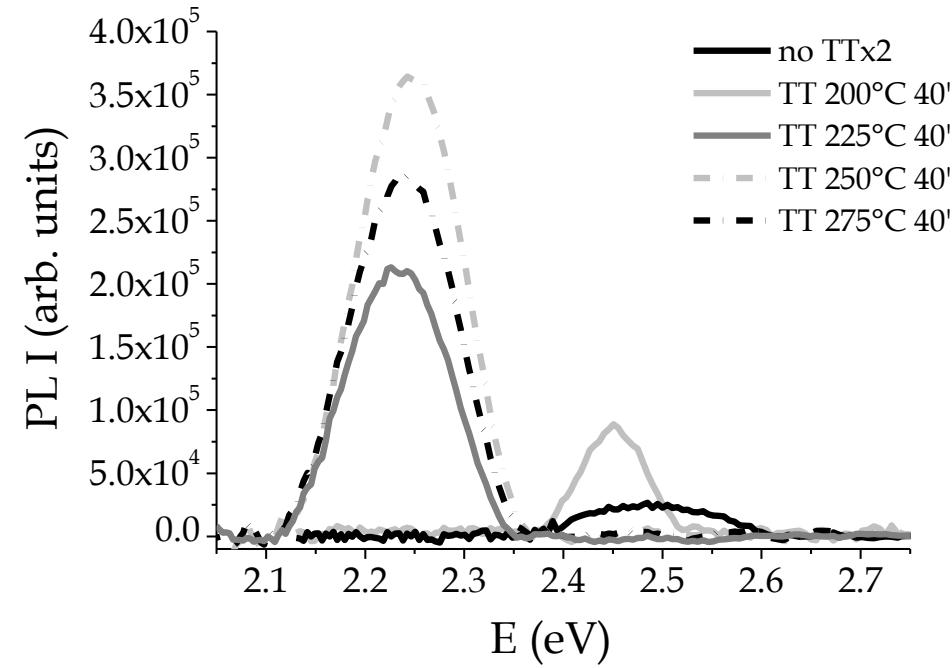
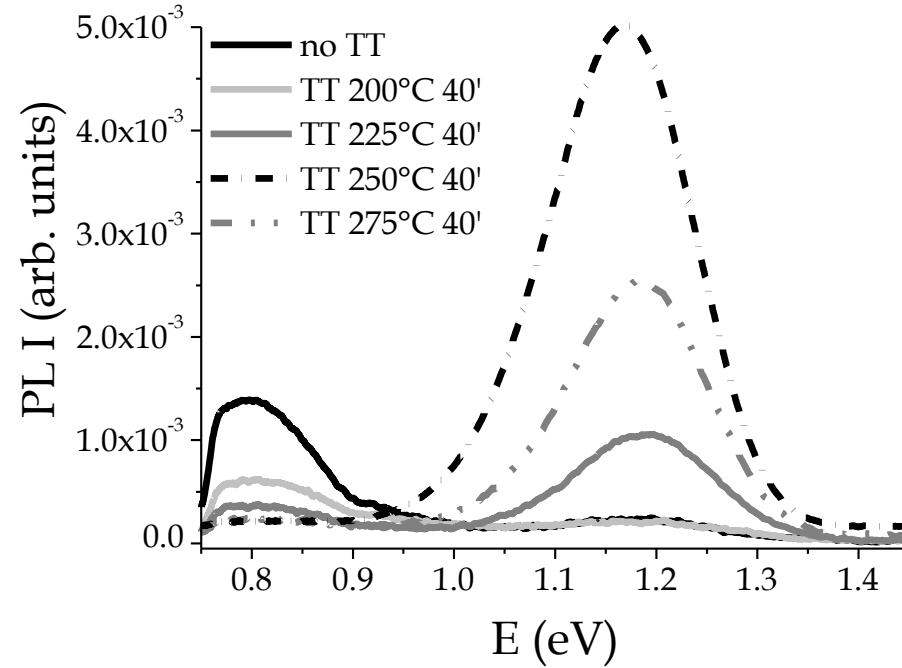


- ✓ annealing at 225°C allows for an improvement of all the device parameters, to 0.83% efficiency

EQE analyses which show a significant increase of the spectral response between 550 and 800 nm for all the tested temperatures, indicating a reduction of recombination losses. But a gradual decrease of EQE in 350 -550 range

It is the present record as Chen et al. (2015) reported 0.49% maximum efficiency on CMTS layers prepared by direct liquid coating

Effect of low temperature annealing on bulk properties



- ✓ low-temperature annealing generally reduces the density of the deep bulk defect responsible for the emission at 0.8 eV, thus reducing recombination losses.
- ✓ before any annealing, CdS shows a very weak PL emission at about 2.5 eV, (nc-CdS) ; After 200 C annealing the PL signal increases and shifts to 2.45 eV indicating an improvement of CdS crystalline quality, and reducing the Eg value (i.e. 2.25 eV).

A. Le Donne, et al. Solar Energy 149 (2017) 125–131, 3283

Conclusion and Future works

- ✓ We have developed an innovative way to achieve high-quality kesterite thin-films suitable for PV
- ✓ CMTS: a new earth abundant material have been tested as PV absorber
- ✓ CZTS by electrochemical approach is under investigation

To be done :

- Reducing harmful defects in CZTS and CMTS
- Testing alternative Buffer layer to replace CdS by ALD (i.e. $Zn_{1-x}Sn_xO_y$ or $Zn(O,Se)$)
- CMTS and CFTS by ink approach on flexible substrate

Final aim : USE a low cost solution process to get a full inorganic earth abundant based multijunction solar cells

Thank you for your attention !

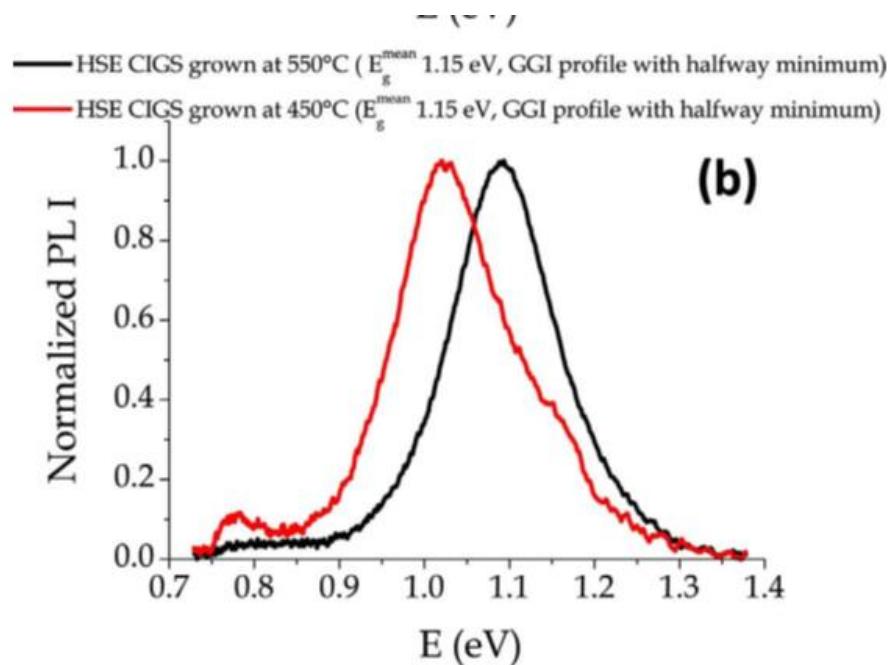


J. Parravicini
A. Le Donne
G. Tserbelidis
V. Trifiletti (now in UK)
M. Acciarri
L. Frioni

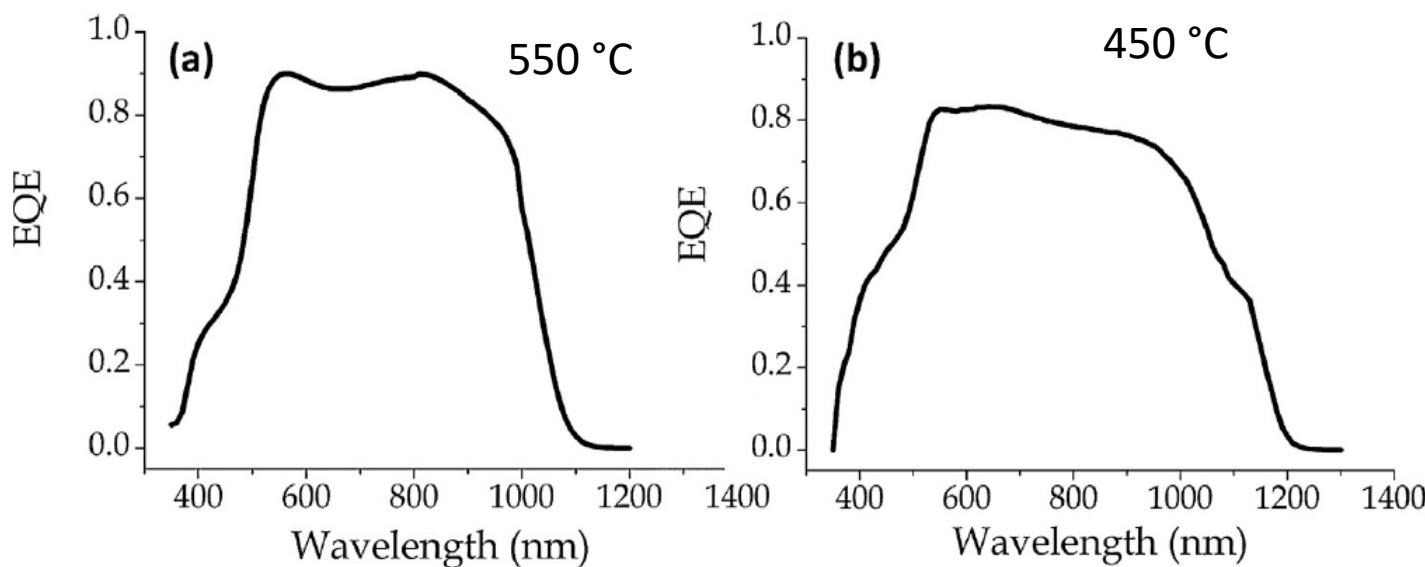
simona.binetti@unimib.it
www.mibsolar.mater.unimib.it

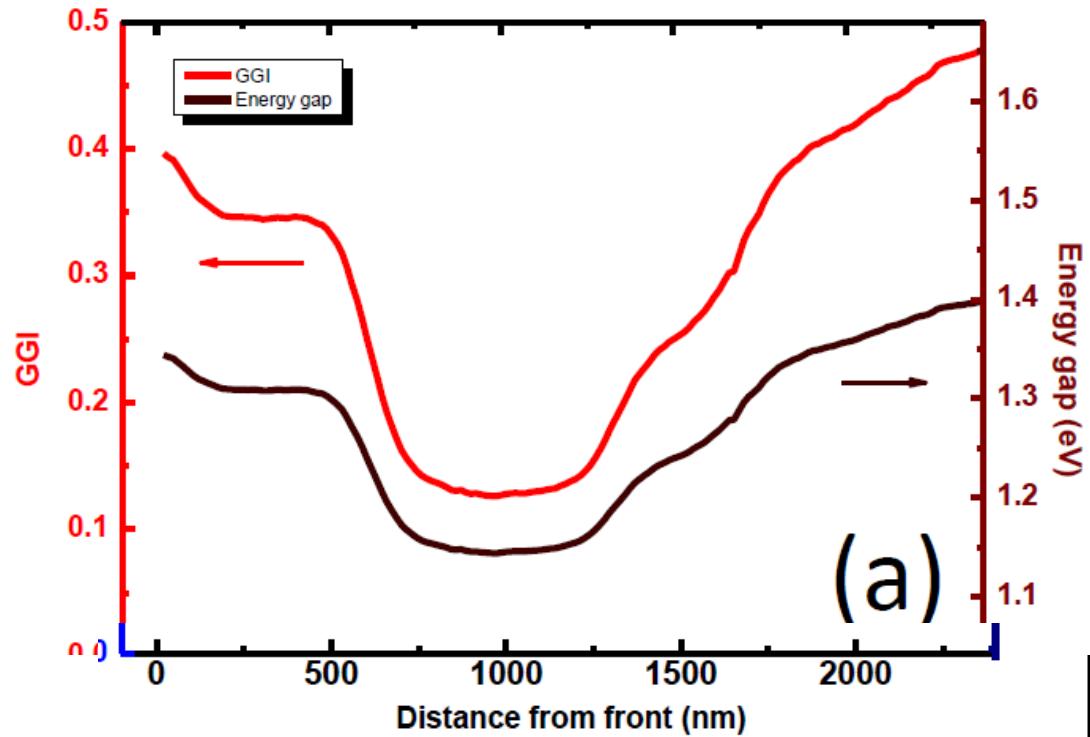
S.Binetti, Sydney 27th November 2019

CIGS

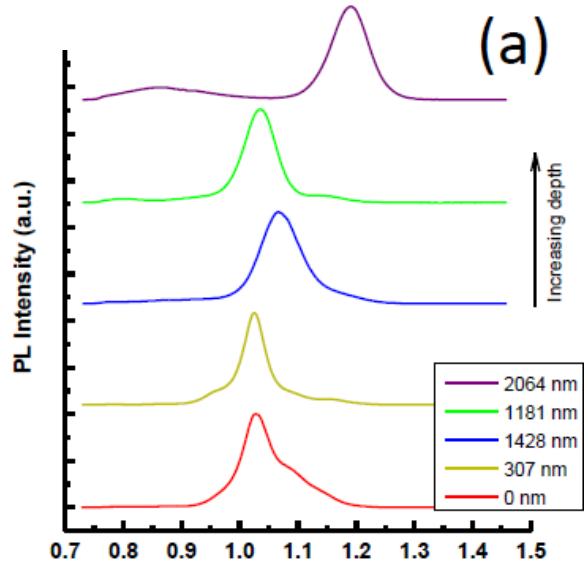


Simulations performed considering the presence of a defective layer at the interface between CIGS and CdS



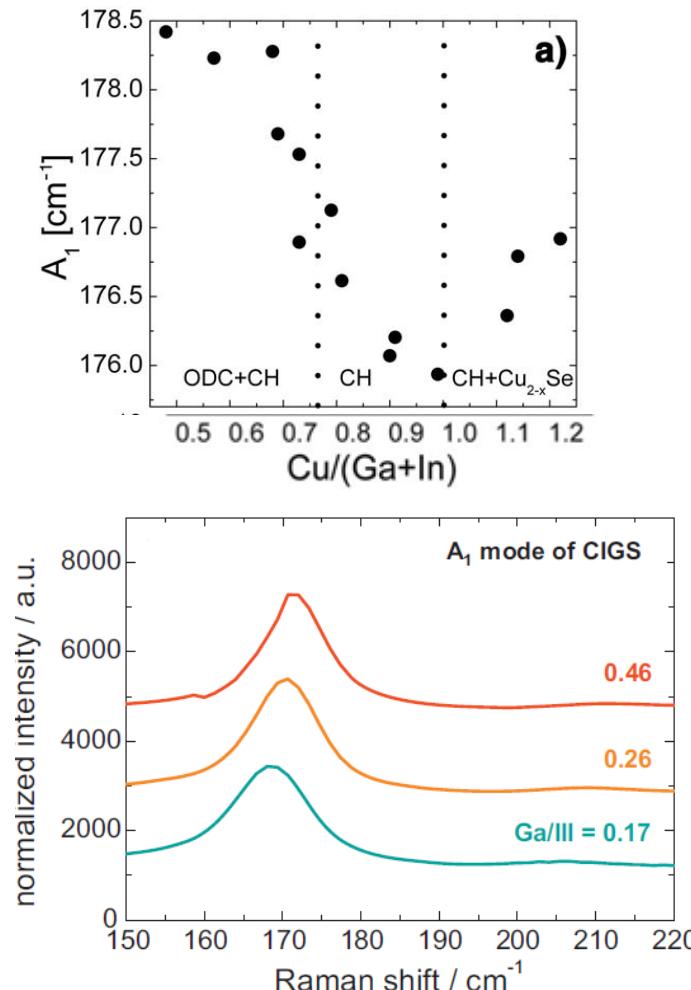
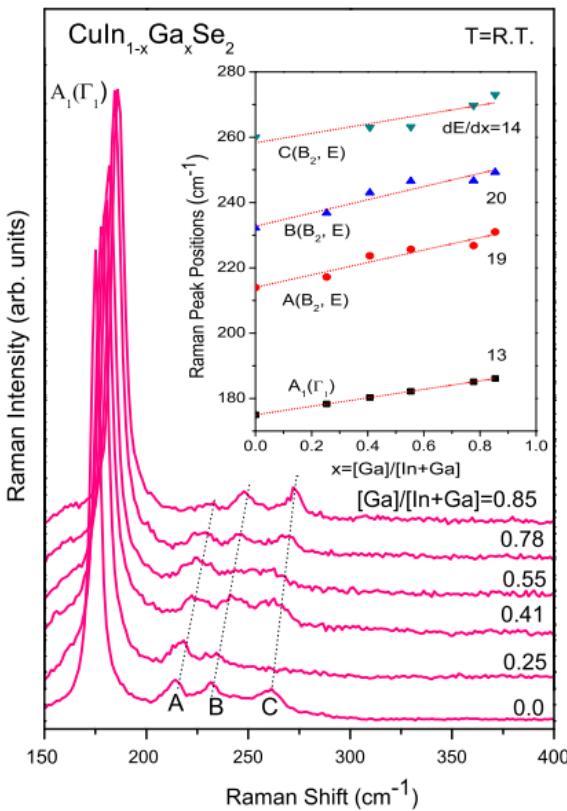


(a)



Raman spectroscopy in CIGS

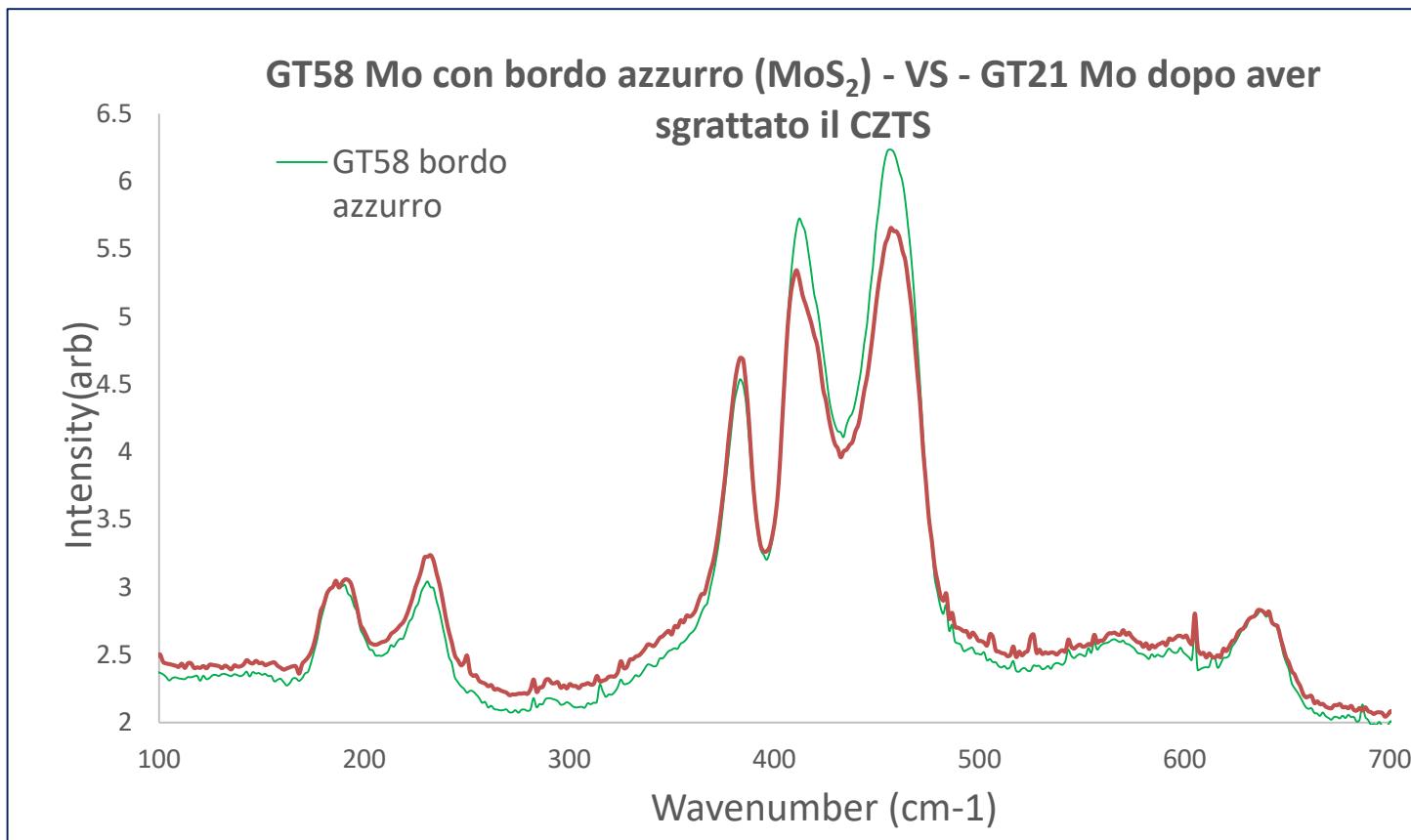
- The main Raman mode A₁ (vibration of the Se anions in the x-y plane with the cations at rest) increases linearly with increasing Ga content (from 174 cm⁻¹ for CIS to 184 cm⁻¹ for CGS): its position can be used to make a quantitative estimation of the mean [Ga]/[Ga]+[In] and its depth gradient – and depends on the relative Cu content
- a shoulder at 150–170 cm⁻¹, is attributed to the OVC (ordered vacancy compound) phase



W. Witte et al . Thin Solid Films
517 (2008) 867–869;

Park et al., J. of Alloy and Comp.
513 (2012)068

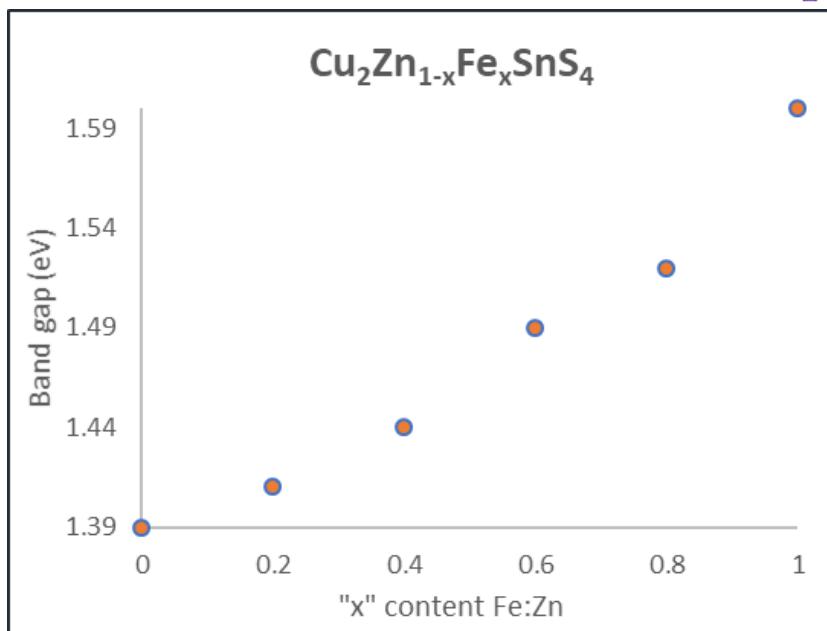
MoS₂ on the back conctac



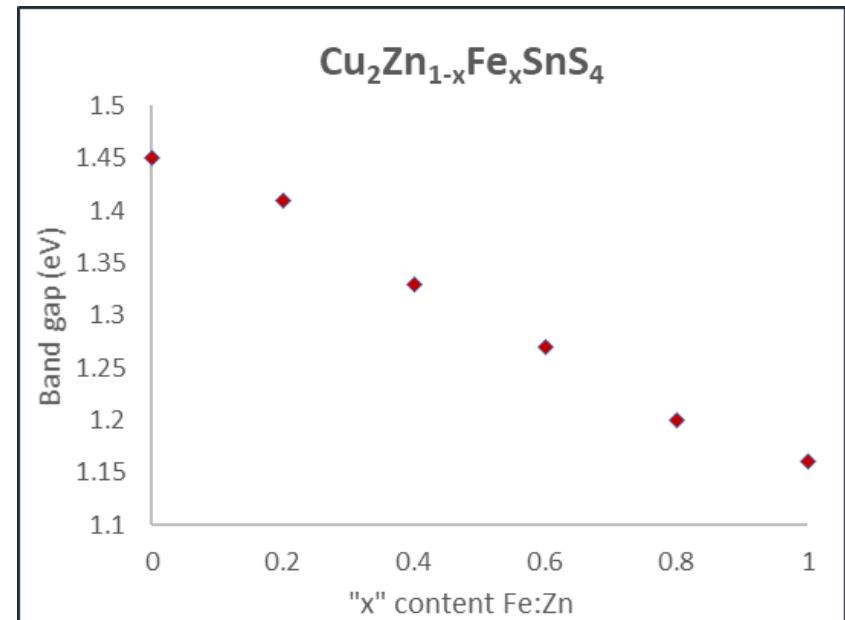
Tuning the gap by $\text{Cu}_2\text{Zn}_{1-x}\text{Fe}_x\text{SnS}_4$



Unpublished results



Working in inert atmosphere



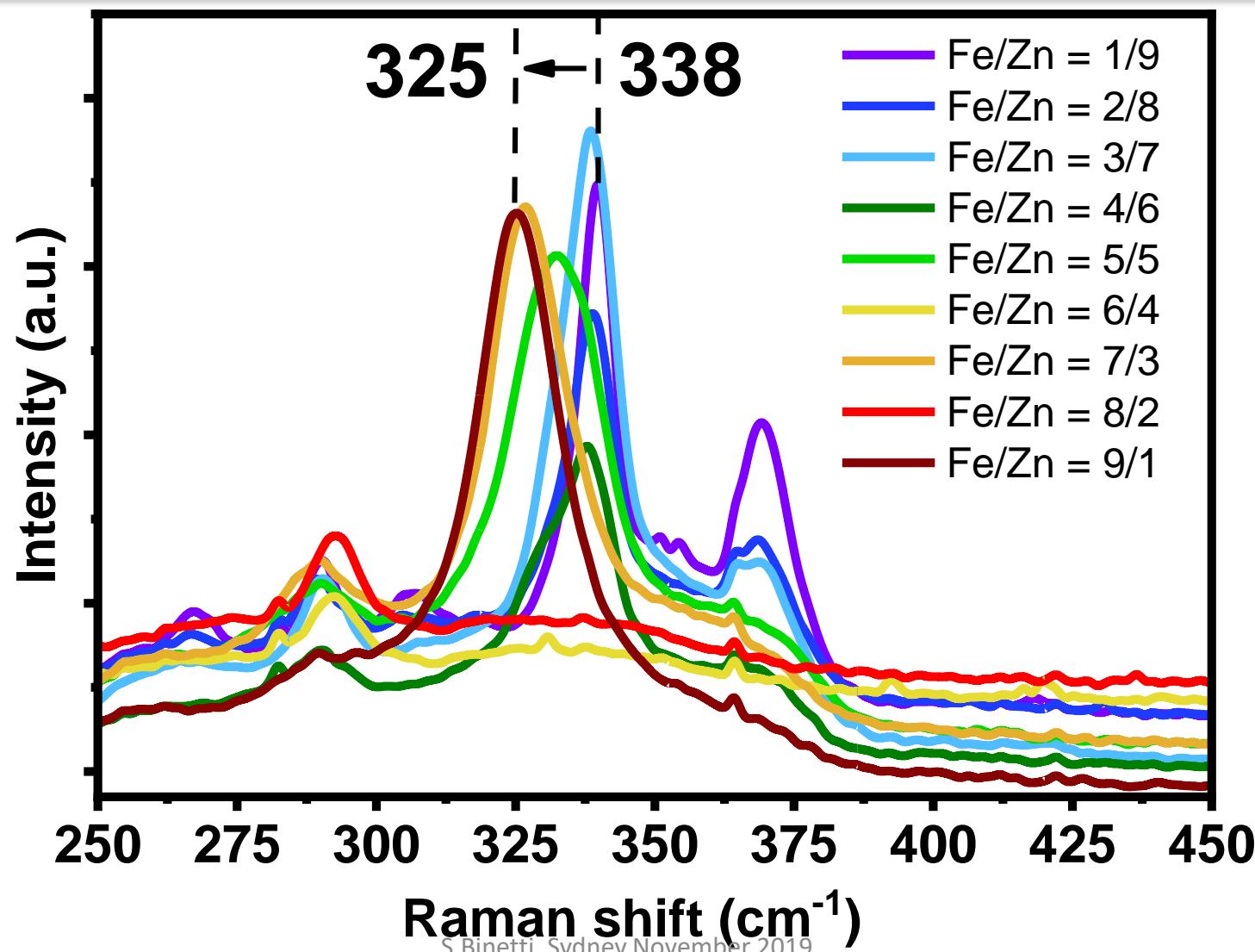
Working in air

S.Binetti, Sydney November 2019

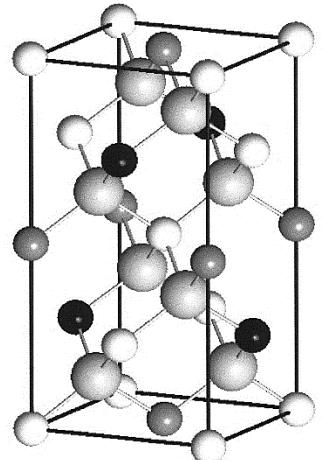
In agreement with J.Phys Chem 118 14227

Gel formation in situ

$\text{Cu}_2(\text{Zn},\text{Fe})\text{SnS}_4$

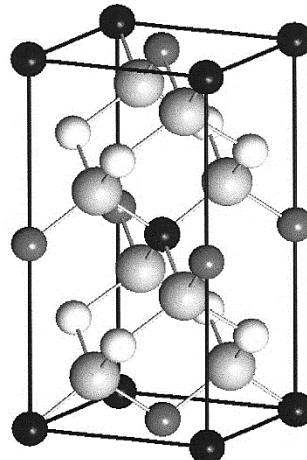


Tuning the gap $\text{Cu}_2\text{Zn}_{1-x}\text{Fe}_x\text{SnS}_4$



Kesterite

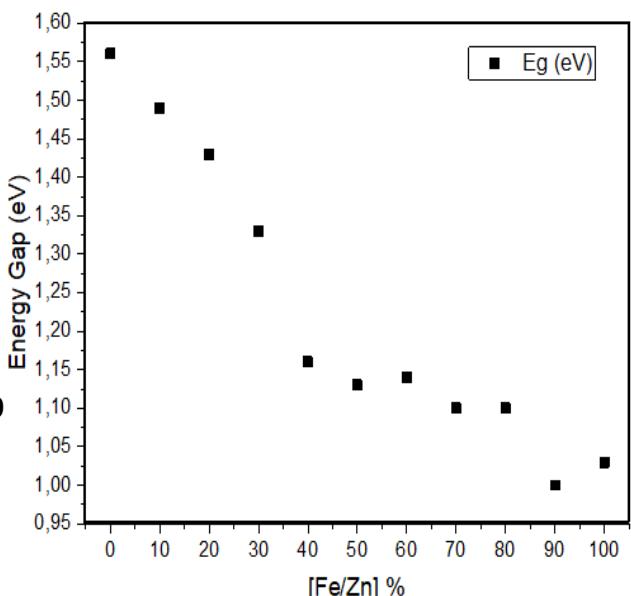
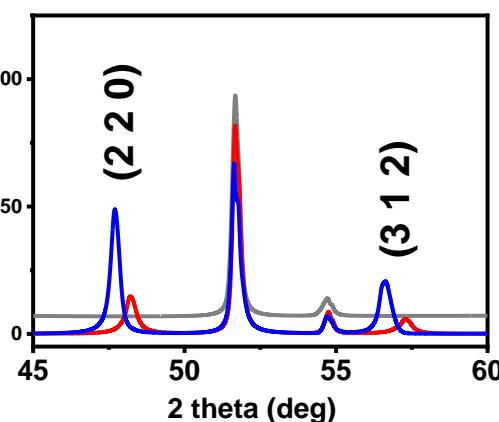
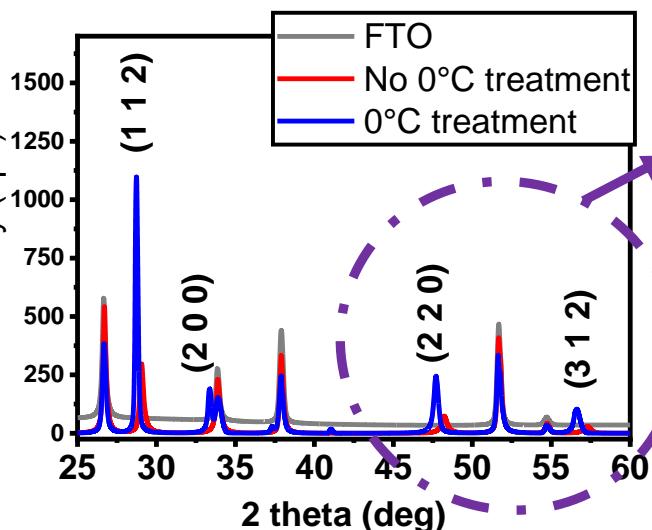
Cu
Sn
Fe,Zn
S



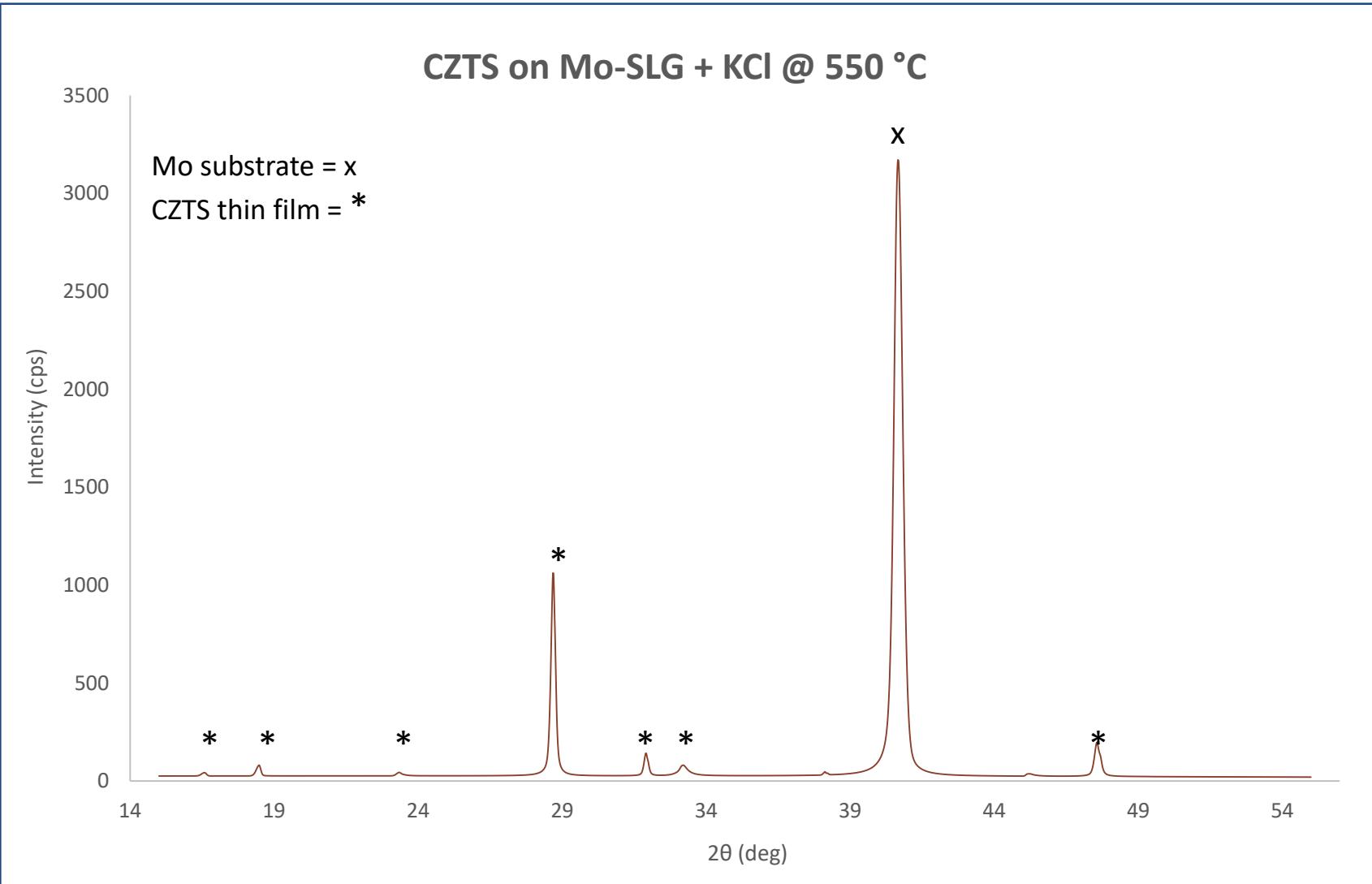
Stannite

$\text{Cu}_2\text{FeSnS}_4$

We were able to grow $\text{Cu}_2\text{FeSnS}_4$ thin-films in kesterite structure, adding a treatment at 0°C after spreading the Ink on FTO.

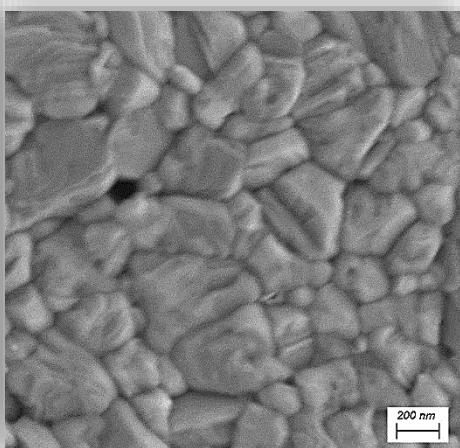
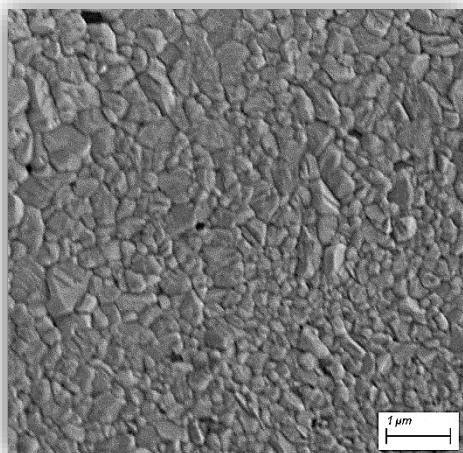
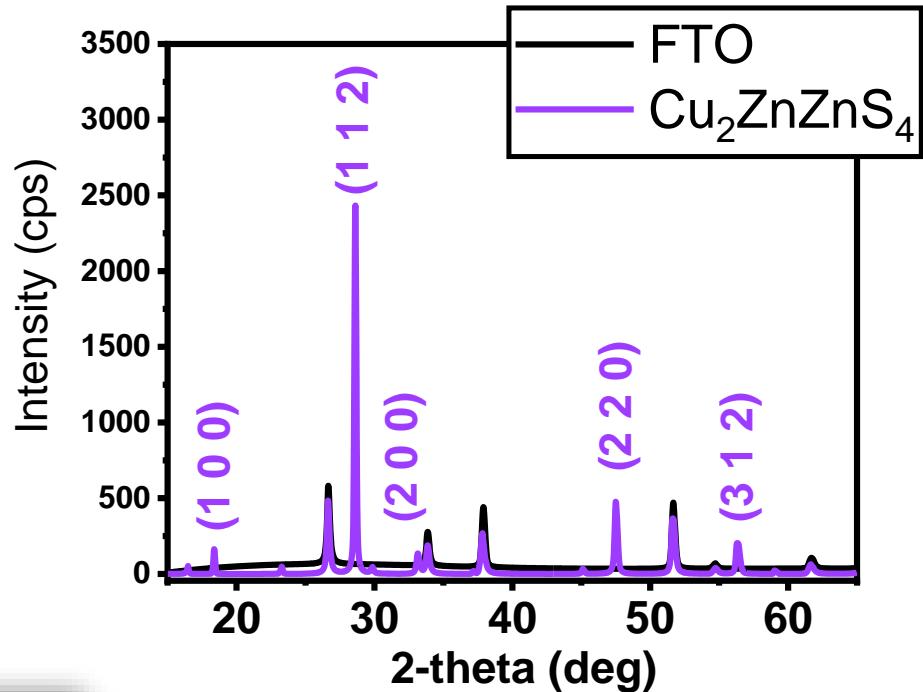
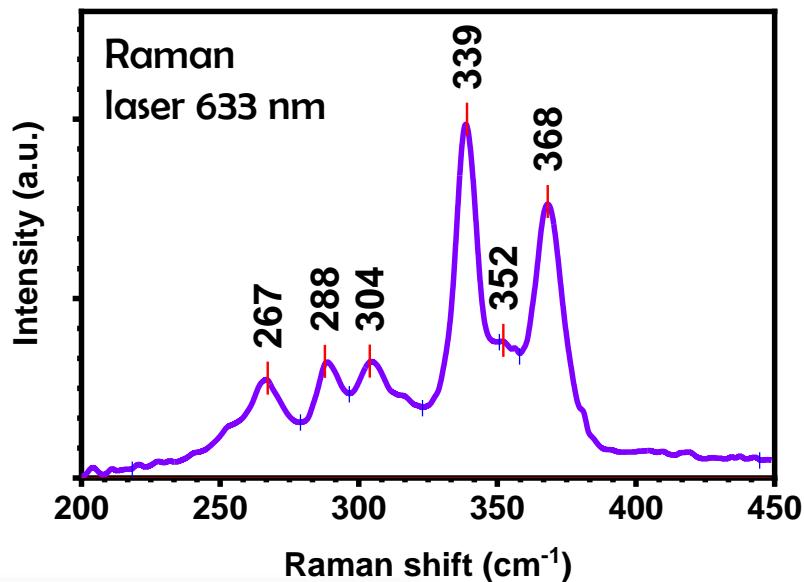


XRD phase analysis:

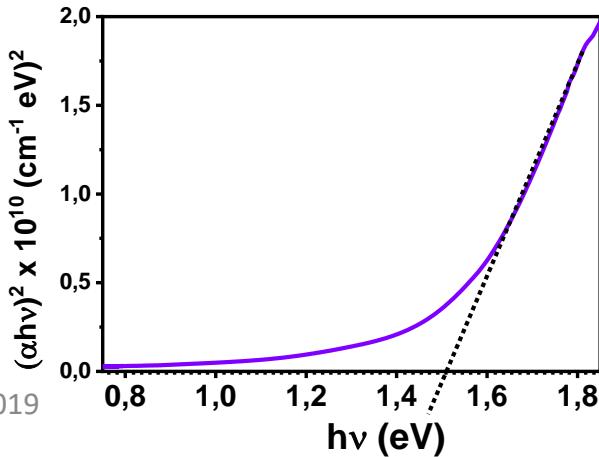


CZTS well-defined peaks with no secondary phases.

Gel formation *in situ* $\text{Cu}_2\text{ZnSnS}_4$



S.Binetti, Sydney November 2019

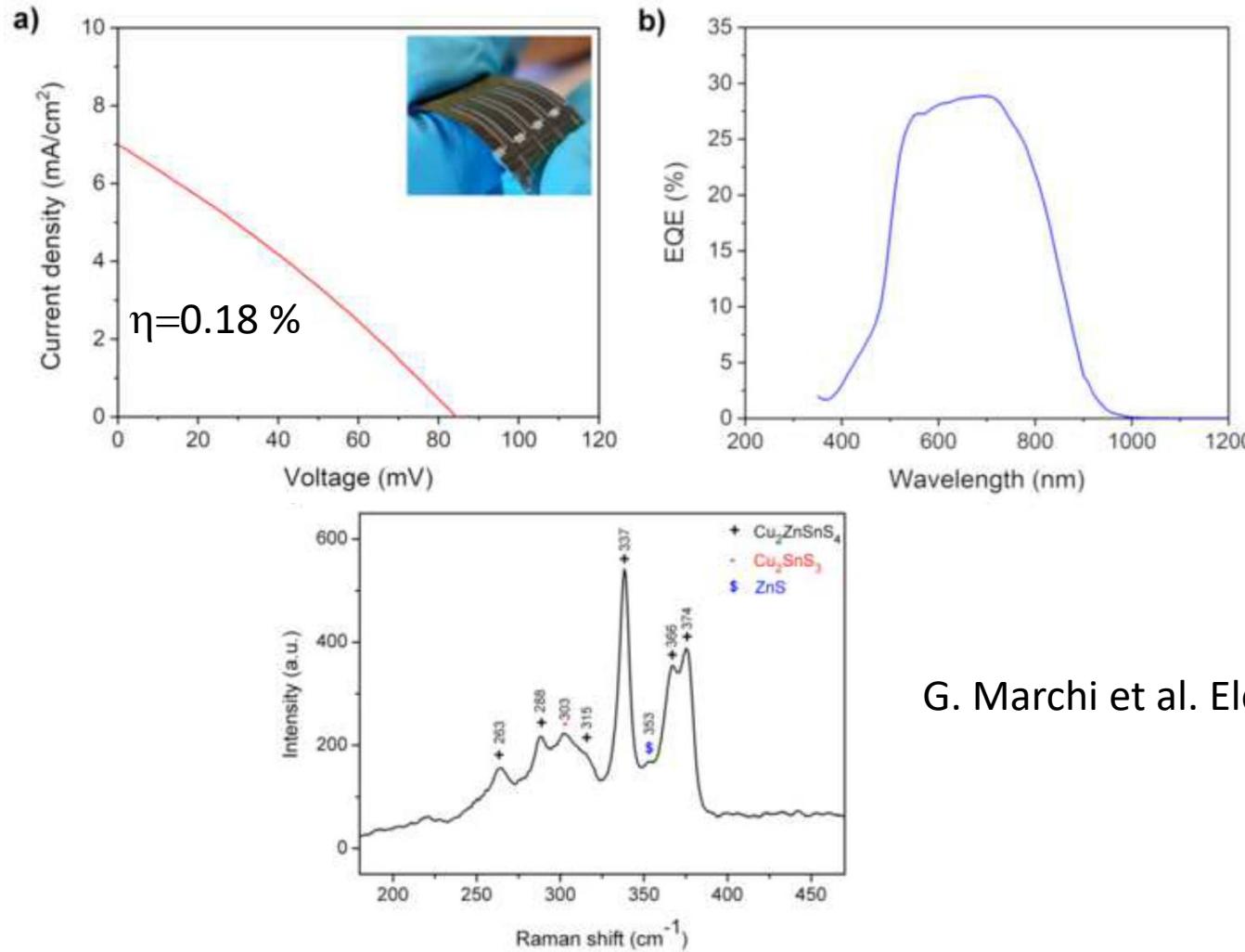


CZTS by electrodeposition

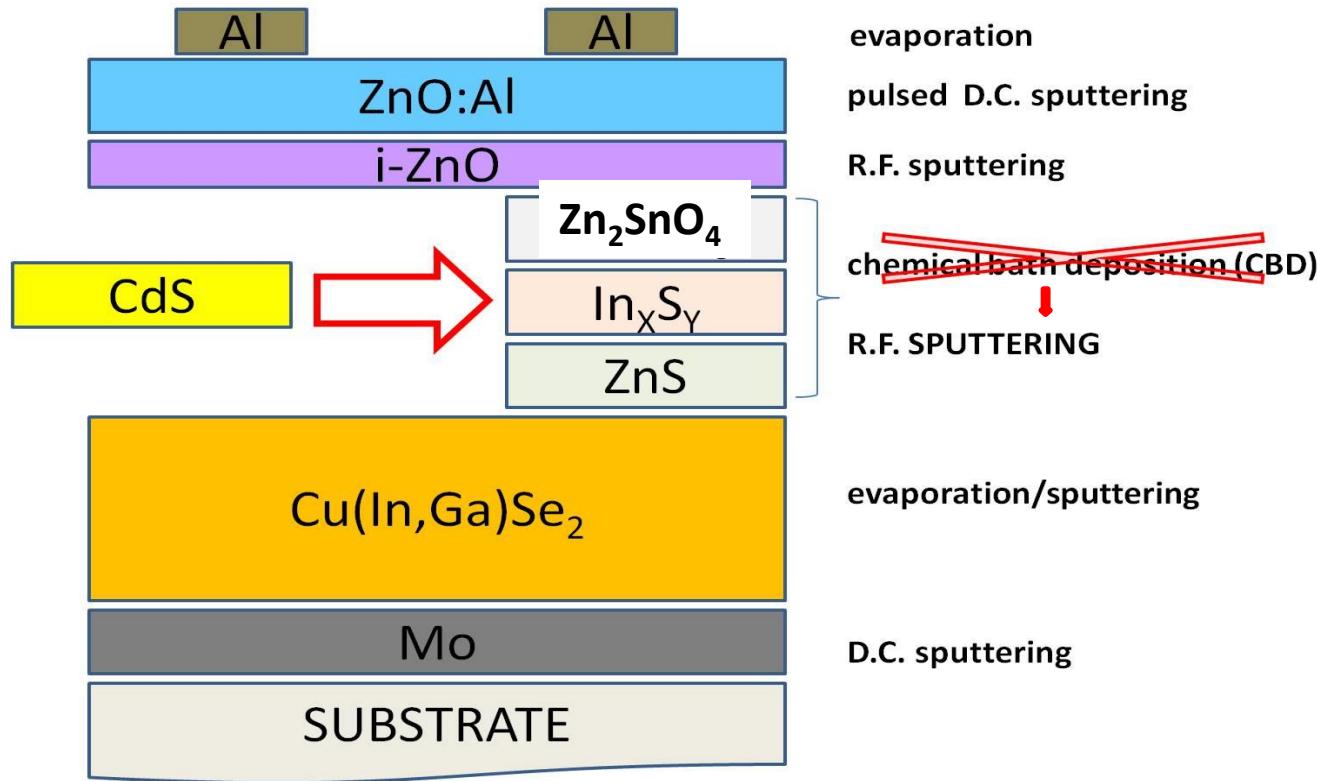


Co-electrodeposition of Cu-Zn-Sn (CZT) from alkaline solution
on flexible Mo substrate followed by sulfurization

Potassium pyrophosphate (K₄P₂O₇), CuCl₂, ZnSO₄·7H₂O, EDTA-Na₂ and SnCl₂·2H₂O
in 100 ml of Millipore water

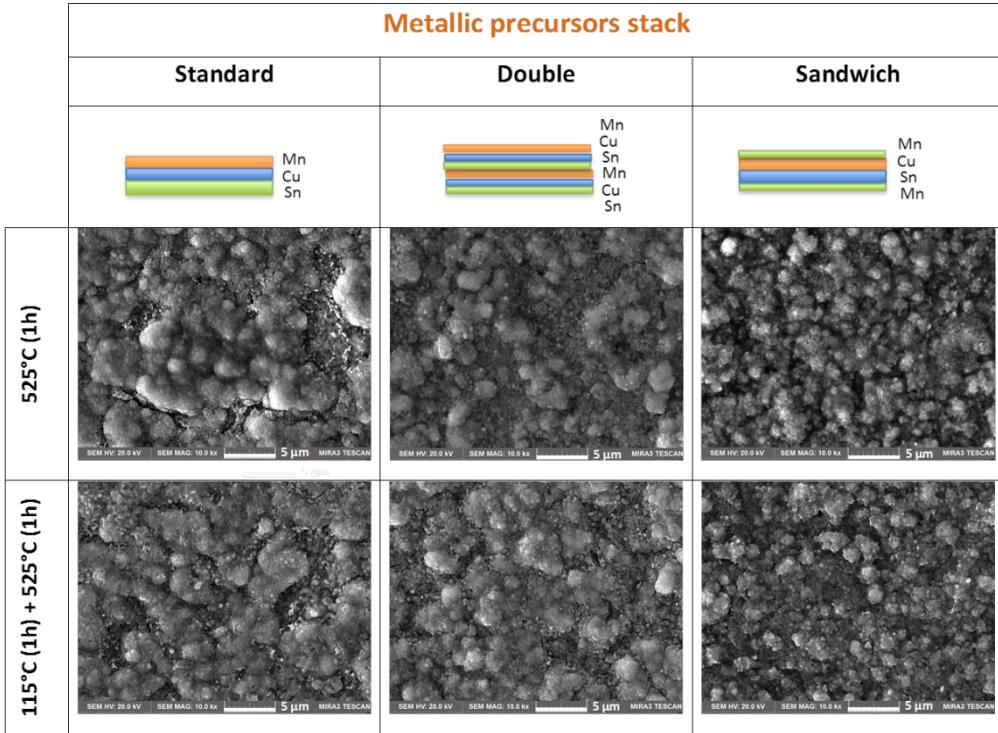


Cd-free alternative buffer layers for Cu(In,Ga)Se₂ and CZTS solar cells

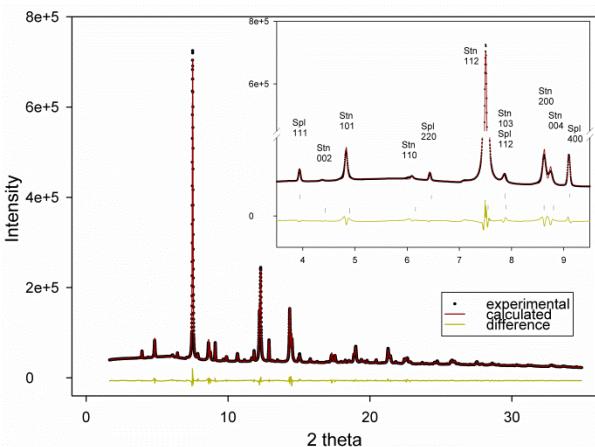


The deposition of the ZTO films was operated both from a ceramic target (75wt% ZnO – 25wt%. SnO₂) and from two metal targets (Zn and Sn) to form a metal bi-layer followed by an oxidation

Cu_2MnSnS_4 : 2nd Series (VS metals sequence and TT @ 115°C)



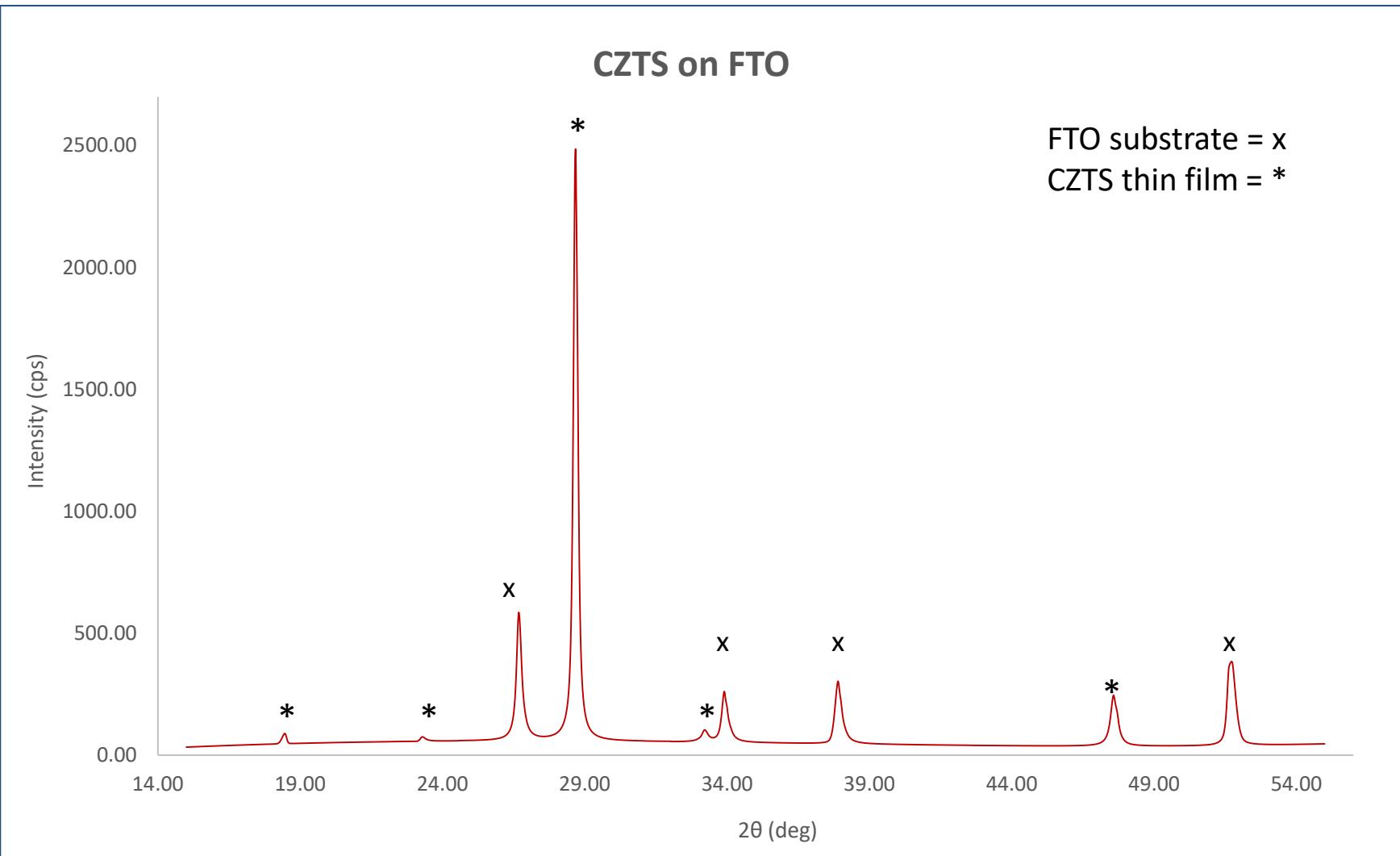
- ✓ the additional **step at 115°C** is mandatory to obtain larger grains, whose size seems to be almost constant for $T > 525^\circ\text{C}$
- ✓ the **sandwich stack** structure provided the best morphology in terms of compactness and grain size, but also the higher content of insulating spinel secondary phase ⇒ the **standard stack** structure provided the best compromise.



Stack structure	Sulfurization temperature	Cell parameters [Å] from XRD analysis			c/a ratio	[Stannite]/[Spinel] ratio
		a	b	c		
Standard	150 +525°C	5.503	5.503	10.855	1.972	89.05%
Double		5.489	5.489	10.823	1.972	87.96%
Sandwich		5.509	5.509	10.841	1.968	69.47%
Standard	150 +550°C	5.505	5.505	10.852	1.971	90.40%
Double		5.507	5.507	10.848	1.970	86.23%
Sandwich		5.511	5.511	10.836	1.966	76.22%
Standard	150 +585°C	5.507	5.507	10.847	1.970	90.71%
Double		5.509	5.509	10.845	1.968	90.15%
Sandwich		5.512	5.512	10.837	1.966	81.90%

Binetti, Sydney November 2019

	<u>Raman Shift (cm⁻¹)</u>
CZTS	266, 288, 338, 368-374
Cu_2SnS_3	295-303, 355
SnS_2	315
SnS	164, 192, 218
ZnS	278, 351
Cu_2S	264, 475

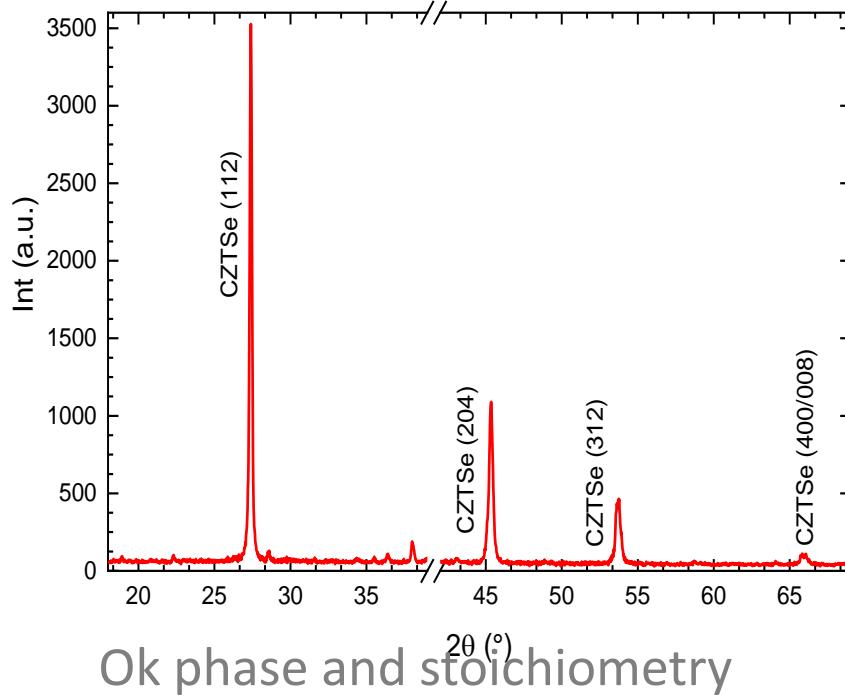


Raman spectrum:



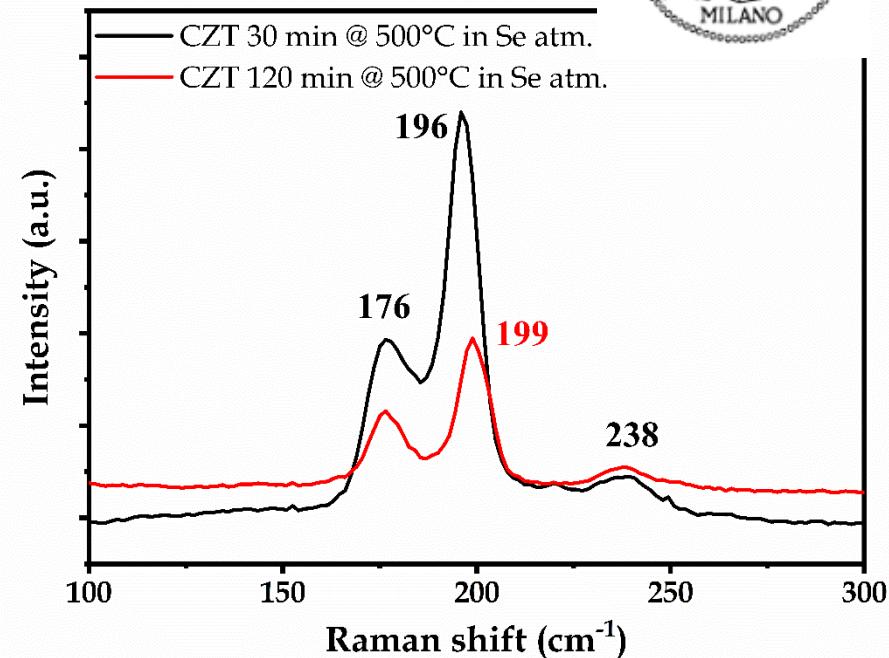
Well-defined peaks, comparable with single crystal XRD!
S.Binetti, Sydney November 2019

CZTSe electrodeposition sample : preliminary results

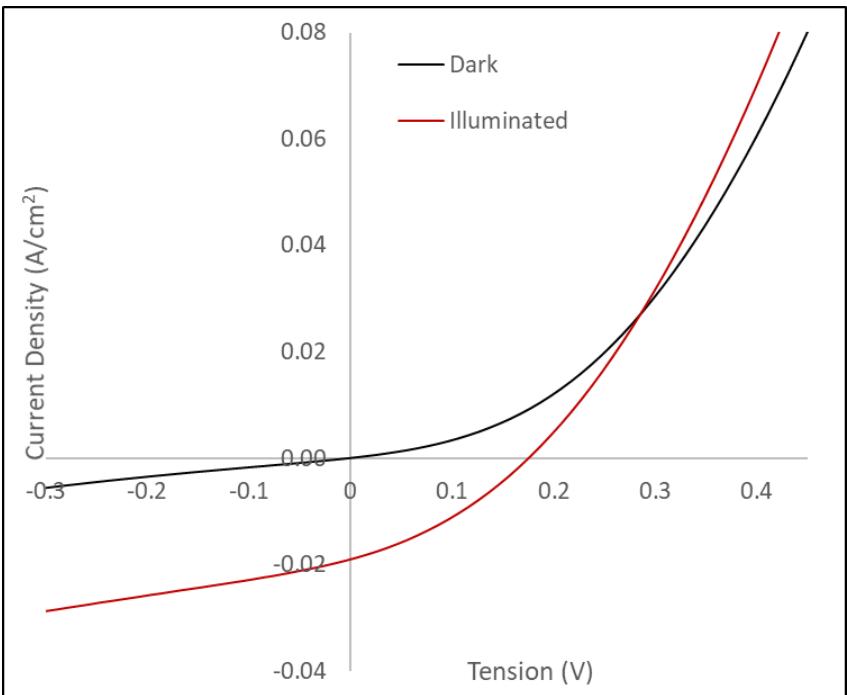


Ok phase and stoichiometry

→ Good homogeneity

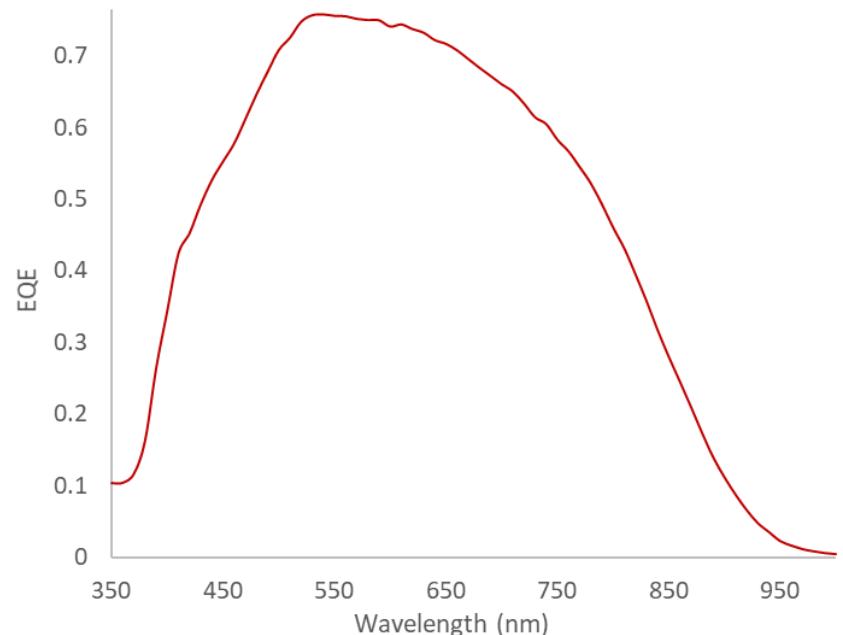
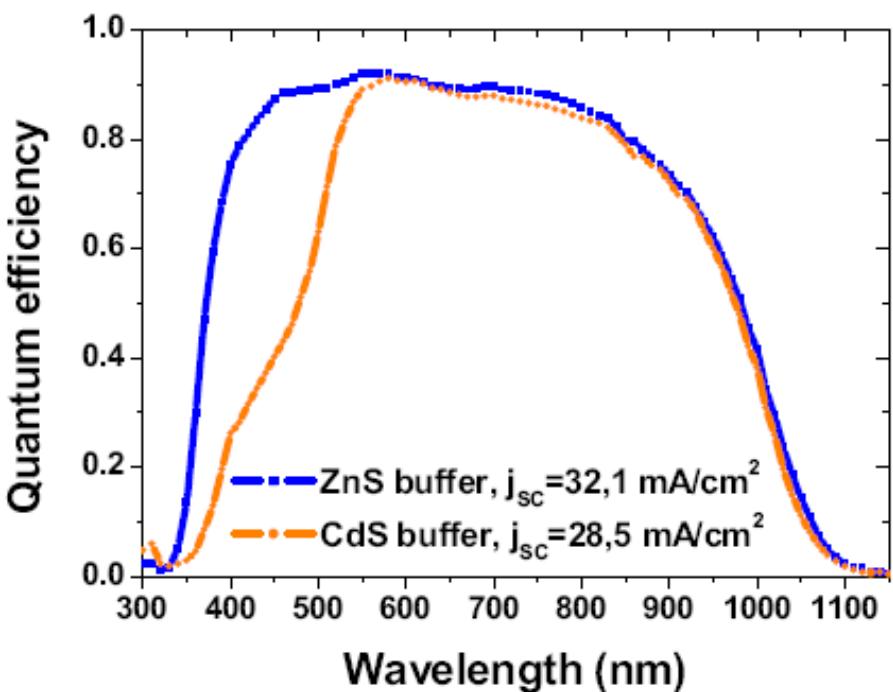


Device Per



$V_{oc} = 175.0 \text{ mV}$ → **Modest V_{oc}**
 $J_{sc} = 18.93 \text{ mA}/\text{cm}^2$ → **Respectable J_{sc}**
 $\eta = 1.11 \%$
 $FF = 33.43 \%$

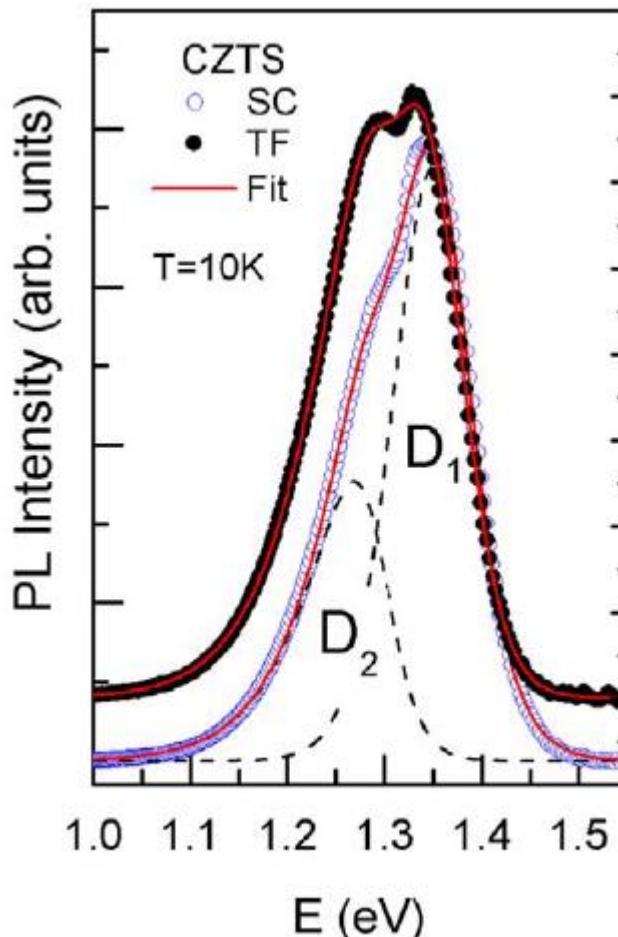
SCAPS software simulation confirms problems at the interface with the back contact



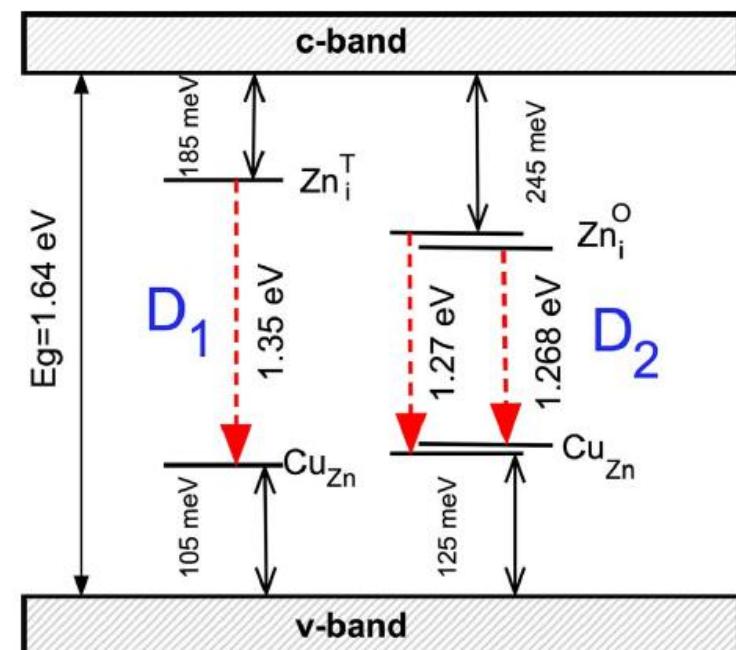
PL results

Togliere

Thin film (TF) compared with Single Crystal (SC) kesterite



These bands could be related to deep donor-deep acceptor (DD-DA) complex defect



The presence of Zn_i must be reduced

J.Krustoka, T.Raadik, M.Grossberg, V. Trifiletti, S.Binetti,
Materials Science in Semiconductor Processing , 80, 52 (2018)

Optimization of the solution

Entry#	Cu/(Sn+Zn)	Zn/Sn	Thiourea [conc.]	Voc (mV)	Jsc (mA/cm ²)	FF %	η %
1	1.00	1	3.7 M	-	-	-	-
2	0.91	1	3.7 M	-	-	-	-
3	0.86	1	3.0 M	-	-	-	-
4	0.83	1	3.0 M	83.0	8.0	29.1	0.18
5	0.80	1	3.0 M	70.0	11.6	26.2	0.21
6	0.80	1.1	3.0 M	149.4	12.6	31.0	0.58
7	0.80	1.2	3.0 M	175.0	18.9	33.4	1.11
8	0.80	1.2	2.3 M	60.0	7.6	27.0	0.12

